

One Step Synthesis and Photocatalytic Activity of Rare Earths/Rutile TiO2

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Rutile TiO₂ and 12 rare earths (RE)/rutile TiO₂ were synthesized by one-step method at 80 °C. Subsequently, they were characterized using XRD, SEM, particle size, DRS and Zeta (ζ) potential. The results indicated that the rare earth/rutile TiO₂ was successfully synthesized at 80 °C and rare earth obviously improved the parameters of photocatalytic activity of rare earth/rutile TiO₂, such as crystalline lattice diameters, particle size, band gap and ζ potential.

Key Words: Rutile TiO₂, One-step method, Rare earth, Low temperature, Photocatalytic activity.

INTRODUCTION

It was evident that the band gap for rutile, which was usually synthesized under high temperature^{1,2} is 3.0 eV. The absorbing wavelength of rutile TiO_2 is much longer than that of anatase TiO_2 (3.2 eV) with a similar crystal grain size³. Zhao *et al.*⁴ synthesized the small grain size and pure rutile TiO_2 using a novel, low temperature based, one-step method. At present, there has been a growing interest in the studying of photocatalytic activity of rutile TiO_2 from a doped materials view⁵.

The influence of rare earth on photo catalytic activity of anatase TiO_2 had been confirmed, several examples of simulation can be found in the literature^{6,7}. Therefore, rare earth improved the band gap of anatase TiO_2 , but it had not been applied to study on rutile TiO_2 . Recently, the band gap and photo catalytic activity of rare earth/rutile had been studied using DFT⁸. However, it has been, until recently, a lack fundamental experimental study on small particle size rare earth/rutile TiO₂ under low temperature.

In the present paper, rare earth/rutile TiO_2 was synthesized using one-step method and the corresponding crystal phase and photocatalytic activity were investigated using the XRD, SEM, particle size, DRS and ζ potential methods^{9,10}.

EXPERIMENTAL

Experimental methods were shown in Fig. 1 and rutile phase appears at (b) step. The volume ratio of $Ti(OC_4H_9)_4$, 2.0

mol L⁻¹ ammonia and 65 % HNO₃ was 1:1:1. The gel was washed by centrifugalization and rare earth oxide in HNO₃ was added into the solution. Therefore, quality ratios of rare earth oxide were 0.0, 0.5, 1.0, 1.5 and 2.0 wt %, respectively. Following vigorous magnetic stirring, the rare earth/rutile TiO₂ was obtained at 80 °C⁴ and the total synthesis time was dominated by its enthalpy of formation. Finally, the rare earth/rutile TiO₂ powders were dried at 90 °C for 2 h.

RESULTS AND DISCUSSION

Synthesis time: Fig. 2 shows that surface morphology of rare earth/rutile TiO_2 depends on its growth ratio and crystalline degree, therefore, its growth ratio is faster than that of rutile TiO_2 . Typically, rare earth improves the crystal degree and grain size of total system⁷.

The synthesis time (t) of rare earth/rutile TiO_2 is dominated by enthalpy of formation (E) of system, in accordance with its calculated data⁸, is directly proportional as eqn. 1. When the E of rare earth/rutile TiO_2 increases, its synthesis time will increase in line.

As depicted in Fig. 3, the E of rare earth/rutile TiO_2 increases with shifting in the atomic number of rare earth. Therefore, the E of La (Y)/rutile TiO_2 is less than that of rutile TiO_2 and is similar to each other, because of the similar outer electrical energy.

$$t = t_{R} + \zeta \cdot (E - E_{R}) / \Delta E \tag{1}$$

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(a) Precipitation
 (b) Peptization
 (c) Drying
 Fig. 1. Flow chart of rare earth/rutile TiO₂ by one-step method. Where the model parameters were based on reported by Bian *et al.*¹⁰. The total energy was calculated using the PBE-GGA method. A kinetic energy cutoff of 350 eV and Fermi energy was chosen for 0-energy point



Fig. 2. Illustration of scanning electron microscope (SEM) to La/rutile TiO₂. It was character by LEOS440, Leica Cambridge LTD company. Large size powder was possibility rare earth/rutile TiO₂ and small powder was rutile TiO₂

where t_R and E_R are the synthesis time and energy of rutile TiO₂, respectively; ζ is the ratio of band gap between DFT simulation and experiment and ΔE is the time and energy separation using orthogonal test.

X ray diffraction studies: As shown in Fig. 3, the crystal phase of different content rare earth/TiO₂ is rutile and the doptant different content is not effective on the crystal phase. According to the Scherrer formula in eqn. 2, the crystalline aberrance increases as follows based on the change in its d-spacing^{3,9,10}.

The crystal plane spacing of rare earth/rutile TiO_2 is altered obviously, according to eqn. 3, includes the master growth surface diameters d_{110} , d_{101} and d_{111} . Its crystalline lattice diameter is shown in Fig. 3 and less than that of rutile TiO_2 . Rare earth is also effective on the (211) and (002) faces, but they are not sensitive to photocatalytic activity⁵.

Typically, the crystalline degree of rare earth/rutile TiO_2 reduces with its E increasing. It depends on surface electronic states of rare earth. Thus there are some miscellaneous peaks on the crystal surface of Gd, Tb, Dy, Tm and Er/rutile TiO_2 .

$$D = K\lambda/\beta \cos\theta \tag{2}$$

$$2d\sin\theta = n\lambda \tag{3}$$

Where K is the Scherrer constant; D is the crystal lattice diameter; β is the integral full width at half maximum. d is the interplanar spacing; n is the diffraction order.

Particle size: The particle size is obtained by the volumeaverage diameter and number-average diameter, according to eqns. 4-5. As shown in Fig. 3, except of La and Y, rare earth causes the agglomeration of rutile $\text{TiO}_2^{9,11}$. The agglomeration of Tb/rutile TiO_2 is the most obviously significant and the agglomeration degree of La (or Y)/rutile TiO_2 is the least.



(a) Different content of La, Ce, Pr, Nd, Sm and Eu /rutile TiO₂



(b) Different content of Gu, 10, Dy, Er, 111 and 1 /rutile 110_2

Fig. 3. XRD patterns of rare earth/rutile TiO₂. It was character by X'PertPro, PANalytical Company, NL. Where rare earth/rutile TiO₂ is tetragonal; its crystal group is P4₂/mmm.

Rare earth improves average diameter of rutile TiO_2 because of increasing E of system. With increasing the doptant content of rare earth⁴, the average diameter of rare earth/rutile TiO_2 increases. This means that rare earth decreases the total particle size, however, increases the local structural agglomeration. Therefore, particle size of 1.5 wt %rare earth/rutile TiO_2 is the least according to Fig. 3.

$$E_{(\theta,w)} = 2\pi A a^2 C J_1(ka\theta) / (ka\theta)$$
(4)

$$\mathbf{I}_{(\theta)} = \mathbf{I}_{a(\theta)} + \mathbf{I}_{b(\theta)} \tag{5}$$

where πa^2 is the hole area; C is the delayed phase position; others are phase position deifference. I is the diffracted intensity; I_a is the vertical polarized light; I_b is the level polarized light.

Diffuse reflection spectroscopy: According to eqn. 6, DRS results include the initial adsorbing position and wavelength removal^{9,10,12}. Therefore, electron translates from $4f^{n}$ (or $4f^{n-1}$) to O2p and adsorbing wavelength increases with reducing E of rare earth/rutile TiO₂. Fig. 4 depicts, with the exception of Ce and Nd, rare earth improves the red shift of the adsorbing wavelength; its band gap is similar to the results using DFT⁴. The corresponding optimized doptant content is 1.5 wt %, its particle size is less than others as shown in Fig. 3.

Typically, the lowest band gap widths of Gd (Tb)/rutile TiO_2 may be too small to be character and some energy levels of Gd (Tb) 4*f* states are barrier by Ti 3*d* states. Their actual experimental results are higher than the estimated data.

$$\mathbf{F}_{(\mathbf{R})} = \mathbf{K} / \mathbf{S} = \varepsilon \mathbf{C} / \mathbf{S} = \mathbf{b} \mathbf{C}$$
(6)

where b is the constant; C is the component concentration; S is the scattering coefficient; K is the absorptivity.

Zeta (ζ) potential: Based on the Helmholtz-Smoluchowski equation of ζ potential, rare earth can improve the surface electric state and adsorbing ability of rutile TiO₂, this is related to the surface electric activity as shown in eqn. 7. When the data of $4f^n$ states (or $4f^{n-1}$ states) and O2p state is greater than 0, the isoelectric point can easily be shifted in the alkaline direction and it is little affect by the doptant content.

As seen in Fig. 5, Pr, Sm, Er and Eu causes double isoelectric points, it increases the uniformity of particle size that is related to surface defects. The rare earth, such as La, Ce, Dy, Gd, Tb, Tm and Y, reduces specific double isoelectric point of rutile TiO_2 to a single-point; it improves the surface adsorption action between organic matters and rutile TiO_2 .

The isoelectric point of Ce (Dy, Gd and Tb)/rutile TiO_2 shifts in the acidic direction and the isoelectric point of La (Tm and Y)/rutile TiO_2 shifts towards the alkaline direction. However, there is no isoelectric point in the plots of ζ potential of Nd/rutile TiO_2 that may be rapid inactivation.

$$\zeta = \eta \Lambda_0 (dEz) / [\varepsilon_0 \varepsilon_r (dAP)]$$
⁽⁷⁾

where ζ is the soltion viscosity; Λ_0 is the electrical conductivity; ϵ_0 is the dielectric constant of free space; ϵ_r is the dielectric constant of electrolyte solution.



(a) Different content of La, Ce, Pr, Nd, Sm and Eu/rutile TiO₂





Fig. 4. Diffuse reflection spectroscopy patterns and average diameters of rare earth/rutile TiO₂. The band gap of rutile TiO₂ is 2.96 eV. Diffuse reflection spectroscopy was character by UV-3150, Shimadzu Company, Japan. And particle size was character by Mastersizer 2000, Malvem Company, UK



(a) Different content of La, Ce, Pr, Nd, Sm and Eu/rutile TiO_2



(b) Different content of Gd, Tb, Dy, Er, Tm and Y/rutile TiO₂

Fig. 5. Plots of ζ potential of rare earth/rutile TiO₂. ζ potential was characterized by 3000HS, Malvem Company, UK

Conclusion

(1) Rutile TiO_2 and rare earth/rutile TiO_2 were synthesized using one-step method at 80 °C; their synthesis times are in direct ratio with calculated data.

(2) Particle size of rare earth/rutile TiO_2 was smaller than that of rutile TiO_2 and its crystalline distortion degree depends on surface electronic states and atomic number of rare earth.

(3) Rare earth improves the adsorbing wavelength and iso-electric point of rutile TiO₂, except Y.

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