

Preparation of Polymer Foam Magnesium Ferrite Nano-Materials by Polymerization Phenomena of Metal Ions in Aqueous Solution

GUOSHENG WANG^{*}, XIAOMENG GENG, JIN CAI and DAN MENG

Department of Chemical Engineering, Shenyang University of Chemical Technology, Shenyang 110142, P.R. China

*Corresponding author: E-mail: wgsh-lyc@163.com

Received: 17 January 2015;	Accepted: 13 March 2015;	Published online: 22 June 2015;	AJC-17362
----------------------------	--------------------------	---------------------------------	-----------

In the present investigation, the polymer foam magnesium ferrite/hematite, $MgFe_2O_4/\alpha$ - Fe_2O_3 and magnesium ferrite ($MgFe_2O_4$) were prepared by using magnesium oxide (MgO) and ferric chloride ($FeCl_3 \cdot 6H_2O$) as starting materials with assisted ultrasonic dispersing technology in the aqueous solution. FTIR, XRD, SEM and TG-DTG-DTA analysis indicated that the transformation of $MgFe_2O_4/\alpha$ - Fe_2O_3 to $MgFe_2O_4$ was obtained by adjusting the molar ratio of Mg/Fe selectively or the pH value of aqueous solution. The grain size and morphology of the resultant nano-crystalline were also strongly affected by the molar ratio of Mg/Fe selectively, the pH value of aqueous solution and the calcinations temperature. The forming mechanism of polymer foam $MgFe_2O_4$ can be explained according to the polymerization phenomena of metal ions in aqueous solution. In this study, not any additives or modifiers were used, which demonstrates the potential application of $MgFe_2O_4$ in various field.

Keywords: Magnesium ferrite, Polymer foam, Hematite, Polymerization phenomena, Metal ions.

INTRODUCTION

Magnesium ferrite (MgFe₂O₄) was regarded as an important candidate of the spinel family which have a cubic structure of normal spinel-type. Applications such as in ferro-fluid, highdensity recording media, heterogeneous catalysis, adsorption, humidity and gas sensors and magnetic technologies *etc*. have been developed as a result of its interesting magnetic and electrical properties combined with chemical and thermal stability¹. Furthermore, the potential applications in drug and bio-molecules carriers, or bio-inspired technology have attracted much attention by coating or layered on the magnesium ferrite core².

Magnesium ferrite (MgFe₂O₄) can be prepared by conventional ceramic method (ball milling)³⁻⁷, polymeric precursor method⁸, sol-gel method⁹⁻¹², oxygen-plasma-assisted molecular beam epitaxy¹³, combustion reaction method¹⁴⁻¹⁸, layered precursor method¹⁹, low-temperature sol-gel auto-combustion method²⁰, co-precipitation method²¹⁻²⁴, electro-spinning method²⁵, supercritical hydrothermal reaction²⁶, a molten salt synthesis method²⁷, a one-pot solvo-thermal route²⁸, PEG assisted micro-emulsion method^{29,30}. Various wet-method have been reported for preparing spinel-type ferrite fine particles, in which magnesium salt included magnesium sulfate, magnesium nitrate and hydrated magnesium acetate *etc*. were often used as a starting materials and some of the organic acids included oleic acid, citric acid, acetic acid and glycine *etc.* so much as surfactant were used as dispersing agent. In additions, some of the alkaline substances included sodium hydroxide, potassium hydroxide, so much as ammonia *etc.* were used as regulator of pH value in aqueous solution³¹⁻³³. However, these surface and pH regulator may be remained in the product, which is unfavourable to the application. Therefore, preparation of MgFe₂O₄ without using any surface and pH regulator is important.

In the present investigation, magnesium oxide (MgO) and ferric chloride (FeCl₃· $6H_2O$) were used as starting materials and ultrasonic assisted dispersing were used. The molar ratio of Mg/Fe was selectively to adjust the pH value of aqueous solution. Not any additives or modifiers were used. The behaviour of the synthesized precursor at different pH value was studied. The grain size and morphology of the resultant nanocrystalline were investigated.

EXPERIMENTAL

The chemicals used in the synthesis of samples were magnesium oxide powder (MgO) and ferric chloride (FeCl₃·6H₂O). Magnesium oxide was dissolved in the distilled water with ultrasonic stirring (frequency 30 Hz and 300 rpm) for 1 h at 50 °C, forming MgO emulsion. FeCl₃·6H₂O was also dissolved in the distilled water in the same condition as mentioned above. Then the FeCl₃ was dropwise added to the prepared MgO emulsion, after the completion of drop, the reaction was continued for 1 h with ultrasonic stirring at 50 °C. Finally, the precursor were filtered, washed and dried at 100 °C for 4 h and then the formed precursor powders were calcined at 600, 800 and 1000 °C respectively for 2 h in a digital furnace. The X-ray diffraction (XRD) patterns of the resulting products were characterized by a Brucker D8-advance X-ray powder diffractometer with CuK_{α} radiation (k = 1.5406 Å) in the 2 h range from 10 to 80°. The micrograph of MgFe₂O₄ was examined by direct observation via scanning electron microscope (SEM) model (JSM-5400). The Fourier transformer infrared (FT-IR) patterns of the resulting products were characterized by a BrukerV-70 infrared spectrometer.

RESULTS AND DISCUSSION

pH value measured: The pH value measured as function of time with different molar ratios of (Mg/Fe) was shown in Fig. 1. It can be seen that the pH value measured decrease with increase of reaction time as molar ratio of (Mg/Fe) above 2.5:1 and finally reach a curtained value. The pH value measured decrease and then suddenly down decrease to a certained value as molar ratio of (Mg/Fe) below 2.25:1. On the other hand, the solution shows basic as molar ratio of (Mg/Fe) above 2.5:1 and acidity below 2.25:1 in the reaction at the end. The pH value measured in the reaction at the end as function of molar ratio of (Mg/Fe) was shown in Fig. 2, the S-type curve was clearly obtained. This phenomenon indicated that the reaction was part of alkali neutralization reaction in fact as Fe³⁺ solution was dropped into the MgO emulsion.

TG-DTG-DTA characterization: In aqueous chemistry, water plays a double role. It behaves as a solvent with a high dielectric constant that favours the dissociation of ionic species as:

$$(\mathbf{M})^{z+} + (\mathbf{Cl})_{n} \xrightarrow{\mathbf{H}_{2}\mathbf{O}} (\mathbf{M})^{z+}_{aq} + (\mathbf{Cl})^{-}_{n,aq}.$$
(1)

where M is metal ion. It also behaves as a σ -donor molecule and reacts as a nucleophlic ligand giving solvated species as:



Fig. 1. pH value measured as function of time with different molar ratios of (Mg/Fe)



Fig. 2. pH value measured as function of molar ratio of (Mg/Fe)

$$(M)(OH_2)_n^{z+}$$
 (2)

when the pH value increase,

$$(\mathbf{M})(\mathbf{OH}_2)_n^{z+} \xrightarrow{\mathbf{OH}^-} [\mathbf{M}(\mathbf{OH})_h(\mathbf{OH}_2)_N]^{(z-h)+}$$
(3)

where h is the hydrolysis ratio.

Here, we consider that the magnesium and ferrite hydrated ion are formed in aqueous solution. These hydrated ions in aqueous solution take place ionic polymerization, forming generally recognized as magnesium or iron oxide octahedral or tetrahedron structure:



Sequentially, the back-to-back connection or cross linking of the magnesium or iron oxide octahedral or tetrahedron structure will formed as a results of share of the hydroxyl ion or the H₂O. The polymerization phenomena of metal ions in aqueous solution took place, the compound of magnesium and iron oxide architecture in nanospace were obtained, then the polymer foam of MgFe₂O₄ will be prepared after calcination.

Thermal decomposition of the precursor with Mg/Fe molar ratio 1.75 (pH = 0.95) consists of two steps as shown in Fig. 3. In the TG-DTA curve of the dried gel (pH = 0.95), the first weight loss represents the water vaporization of O-H groups and the second weight loss corresponds with the decomposition of Cl⁻ ions. Therefore, it is confirmed that the amount of heat produced depends on the hydrolysis temperature and time, which is a key factor affecting the size and morphology of the particles.

X-ray analysis: The XRD patterns of the powders heattreated at 800 °C with different Mg/Fe molar ratios from 1.25 to 4 are shown in Fig. 4. The diffraction peaks at 2 θ values of 30.1, 35.5, 43.1, 53.5, 57.1 and 62.6, are corresponding to (220), (311), (400), (422), (511) and (440) planes, respectively,



Fig. 3. TG-DTG-DTA curves of precursor sample at Mg/Fe molar ratio 1.75:1

which provide a clear evidence for the formation of the spinel structure of the magnesium ferrites as Mg/Fe molar ratio 4:1. The diffraction peaks at 24.1, 33.1, 35.6, 40.8, 49.5, 54.1, 57.6, 62.5 and 64.0, are corresponding to (012), (104), (110), (113), (024), (116), (122), (214), (300), (010) and (220) planes of



Fig. 4. X-ray diffraction patterns of MgFe₂O₄ samples at different Mg/Fe molar ratios

rhombohedral α -Fe₂O₃ and (220), (400) and (511) *etc.*, corresponding to planes of the spinel structure of the magnesium ferrites existed and increased as Mg/Fe molar ratio from 2.5:1 to 3.0:1. And only (311), (422) and (440) coexisted as Mg/Fe molar ratio from 1.25:1 to 2.25:1. These results show clearly that the transformation of MgFe₂O₄ / α -Fe₂O₃ to MgFe₂O₄ as Mg/Fe molar ratio from 1.25:1 to 4:1.

SEM observation: Fig. 5 shows the SEM images of $MgFe_2O_4/\alpha$ -Fe_2O₃ after calcinations at 800 °C for 2 h. It can



Fig. 5. SEM images of MgFe₂O₄/ α -Fe₂O₃samples at different Mg/Fe molar ratios (a: 1.25:1, b: 1.5:1, c: 1.75:1, d: 2.0:1, e: 2.25:1, f: 2.5:1, g: 2.75:1, h: 3.0:1, i: 4.0:1)

be seen that the particles are uniformly and the diameter of the particles increase as Mg/Fe molar ratio from 1.25:1 to 1.75:1. Some of the composite appear rough on the surface as Mg/Fe molar ratio from 2.0:1 to 2.5:1. The uniformly siamesed particles can clearly be observed as Mg/Fe molar ratio from 2.75:1 to 3.0:1. The micrographs show the presence of particles that are agglomerated together as Mg/Fe molar ratio 4.0:1. These results show that the grain size and morphology are strongly affected by the Mg/Fe molar ratios.

FT-IR study: The FTIR spectrum of synthesized powder of MgFe₂O₄/ α -Fe₂O₃ in range 4000-350 cm⁻¹ up to down as Mg/Fe molar ratio from 1.25:1 to 2.75:1 was shown in Fig. 6. The broad bands centered around 3420 cm⁻¹ was associated with the O-H stretching vibration of the hydroxyl groups in the brucite layers and interlayer water molecules. The water deformation band was observed at 1631 cm⁻¹. Two persistent absorption bands of metal-oxygen bond corresponding to the vibration of tetrahedral and octahedral complexes at 588 and 482 cm⁻¹, respectively, are observed, which is the evidence for the formation of spinel ferrite structure as Mg/Fe molar ratio from 2.5:1 to 2.75:1. It is noteworthy that the values of MgFe₂O₄/ α -Fe₂O₃ are increased compared with single MgFe₂O₄ phase (at 540 cm⁻¹, 476 cm⁻¹). This is due to the redistribution or inversion of cations among A- and B-site after the incorporation of MgFe₂O₄ and α -Fe₂O₃. The bands between 1421 cm⁻¹ and 1120 cm⁻¹ are assigned to the vibrational bands of C-H bending frequencies or Cl⁻ ion.



Fig. 6. FT-IR spectrum of synthesized powders of MgFe₂O₄/α-Fe₂O₃ at different Mg/Fe molar ratios

Calcinations: The SEM images of $MgFe_2O_4/\alpha$ -Fe_2O_3 samples calcined at 1000 °C when the Mg/Fe molar ratio is 1.25:1, 1.5:1 and 1.75:1and the XRD patterns of $MgFe_2O_4$

 α -Fe₂O₃ samples calcined at 600, 800 and 1000 °C when the Mg/Fe molar ratio is 1.25:1, 1.5:1 and 1.75:1 were shown in Figs. 7 and 8. It can be seen that the diameter increases obviously and the morphology of the as made samples at 1000 °C approach a approximate polygon construction. The intensity of absorption peaks increased gradually with increase



Fig. 8. XRD patterns of MgFe₂O₄ samples calcined at 600, 800 and 1000 °C when the Mg/Fe molar ratio is 1.25:1, 1.5:1 and 1.75:1



Fig. 7. SEMimages of MgFe₂O₄ samples calcined at 1000 °C when the Mg/Fe molar ratio (j: 1.25:1, k: 1.5:1, l: 1.75:1)



Fig. 9. SEM images of MgFe₂O₄/ α -Fe₂O₃ samples calcined at 800 °C when the Mg/Fe molar ratio (m: 1.25:1, n: 1.75:1, o: 2.0:1)

of Mg/Fe molar ratio from 1.25:1 to 1.75:1. It is confirmed the transformation of the MgFe₂O₄/ α -Fe₂O₃ phases in Fig. 8.

Polymer foam: The SEM images of MgFe₂O₄/ α -Fe₂O₃ samples calcined at 800 °C when the Mg/Fe molar ratio is 1.25:1, 1.75:1 and 2.0:1 were shown in Fig. 9 (m, n, o), respectively. It can be seen that perfect polymer foam have been obtained. The particle diameter increases with the increase of Mg/Fe molar ratio from 1.25:1 to 2.0:1. On the other hand, proves that the polymerization phenomena of metal ions in aqueous solution increase with the increase of pH value, or, metals irons transform from low polymer to high cohesion with Mg/Fe molar ratio change from 1.25:1 to 2.0:1.

VSM study: Hysteresis loop of MgFe₂O₄ samples calcined at 800 °C prepared at the Mg/Fe molar ratio from 1.25:1 to 3.0:1 are shown in Fig. 10. It is clearly seen that the value of saturation magnetization (Ms) of the sample prepared at the Mg/Fe molar ratio from 1.25:1 to 2.25:1 is smaller than that prepared at the Mg/Fe molar ratio from 2.5:1 to 3.0:1, the variation in magnetization could be understood as a result of the formation of the spinel structure of the magnesium ferrites prepared at the Mg/Fe molar ratio from 2.5:1 to 3.0:1 and the value of saturation magnetization (Ms) of the sample increase with the Mg/Fe molar ratio from 2.5:1 to 3.0:1. It is also found



Fig. 10. Hysteresis loop of $MgFe_2O_4$ samples calcined at 800 °C prepared at different molar ratio of Mg/Fe

that the coercivity (H_c) of the sample prepared at the Mg/Fe molar ratio from 1.25:1 to 2.25:1 is smaller than that prepared at the Mg/Fe molar ratio from 2.5:1 to 3.0:1, the change in coercivity may be attributed to the change in size of the ferrite particle, as particle size reached the critical size the coercivity go to zero and the grain becomes super-paramagnetic.

Conclusion

The magnesium ferrite/hematite, $MgFe_2O_4/\alpha$ -Fe_2O₃ and magnesium ferrite, MgFe₂O₄ were prepared by using magnesium oxide (MgO) and ferric chloride (FeCl₃·6H₂O) as starting materials with assisted ultrasonic dispersing technology in the aqueous solution. The transformation of MgFe₂O₄/ α -Fe₂O₃ to MgFe₂O₄ was obtained by adjusting the molar ratio of Mg/Fe selectively or the pH value of aqueous solution. The grain size and morphology of the resultant nano-crystaline are strongly affected by the Mg/Fe selectively or the pH value of aqueous solution and the calcinations temperature. The forming mechanism of polymer foam MgFe₂O₄ can be explained according to the polymerization phenomena of metal ions in aqueous solution. In this study, not any additives or modifiers were used, demonstrating the potential application of MgFe₂O₄ in various field. The value of saturation magnetization (Ms) of the sample prepared at the Mg/Fe molar ratio from 1.25:1 to 2.25:1 is smaller than that prepared at the Mg/Fe molar ratio from 2.5:1 to 3.0:1, the coercivity (H_c) of the sample prepared at the Mg/Fe molar ratio from 1.25:1 to 2.25:1 is also smaller than that prepared at the Mg/Fe molar ratio from 2.5:1 to 3.0:1.

ACKNOWLEDGEMENTS

This work was supported financially by the Liaoning Province Nature Science Foundation (No. 201202175).

REFERENCES

- N. Mongia, A.K. Srivastava and D. Bansal, Effect of pH on Magnetic and Structural Properties of Low Temperature Synthesized MgFe₂O₄ Nanoparticles, ICANN, pp. 394-400 (2009).
- P.P. Hankare, V.T. Vader, N.M. Patil, S.D. Jadhav, U.B. Sankpal, M.R. Kadam, B.K. Chougule and N.S. Gajbhiye, *Mater. Chem. Phys.*, 113, 233 (2009).
- 3. Y. Shen, Q. Zhao, X. Li, Y. Hou and G. Chen, *Colloids Surf. A*, **403**, 35 (2012).
- 4. D. Chen, Y.Z. Zhang and C.J. Tu, Mater. Lett., 82, 10 (2012).
- M.J. Iqbal, Z. Ahmad, T. Meydan and Y. Melikhov, J. Magn. Magn. Mater., 324, 3986 (2012).
- 6. J. Shah and R.K. Kotnala, Sens. Actuators B, 171-172, 832 (2012).
- J. Shah, M. Arora, L.P. Purohit and R.K. Kotnala, *Sens. Actuators A*, 167, 332 (2011).
- V. Sepelák, M. Menzel, K.D. Becker and F. Krumeich, *J. Phys. Chem. B*, 106, 6672 (2002).

- Y.M.Z. Ahmed, E.M.M. Ewais and Z.I. Zaki, J. Alloys Comp., 489, 269 (2010).
- S. Maensiri, M. Sangmanee and A. Wiengmoon, *Nanoscale Res. Lett.*, 4, 221 (2009).
- S.K. Pradhan, S. Bid, M. Gateshki and V. Petkov, *Mater. Chem. Phys.*, 93, 224 (2005).
- 12. A. Franco Jr. and M.S. Silva, J. Appl. Phys., 109, 07B505 (2011).
- C.-P. Liu, M.-W. Li, Z. Cui, J.-R. Huang, Y.-L. Tian, T. Lin and W.-B. Mi, J. Mater. Sci., 42, 6133 (2007).
- 14. X. Xiang, G. Fan, J. Fan and F. Li, J. Alloys Comp., 499, 30 (2010).
- 15. M.J. Iqbal, Z. Ahmad, T. Meydan and Y. Melikhov, J. Appl. Phys., 111, 033906 (2012).
- 16. L.B. Kong, Z.W. Li, G.Q. Lin and Y.B. Gan, Acta Mater., 55, 6561 (2007).
- R.K. Kotnala, J. Shah, B. Singh, H. Kishan, S. Singh, S.K. Dhawan and A. Sengupta, *Sens. Actuators B*, **129**, 909 (2008).
- 18. S. Darshane and I.S. Mulla, Mater. Chem. Phys., 119, 319 (2010).
- T. Sasaki, S. Ohara, T. Naka, J. Vejpravova, V. Sechovsky, M. Umetsu, S. Takami, B. Jeyadevan and T. Adschiri, *J. Supercrit. Fluids*, 53, 92 (2010).
- 20. S.F. Mansour, J. Magn. Magn. Mater., 323, 1735 (2011).
- 21. A. Franco, V.S. Zapf, V.B. Barbeta and R.F. Jardim, J. Appl. Phys., 107, 073904 (2010).
- 22. A.N. Ay, D. Konuk and B. Zümreoglu-Karan, *Mater. Sci. Eng. C*, **31**, 851 (2011).

- M.J. Iqbal, Z. Ahmad, Y. Melikhov and I.C. Nlebedim, J. Magn. Magn. Mater., 324, 1088 (2012).
- T. Bala, C.R. Sankar, M. Baidakova, V. Osipov, T. Enoki, P.A. Joy, B.L.V. Prasad and M. Sastry, *Langmuir*, 21, 10638 (2005).
- 25. W.H. Butler, X.G. Zhang, T.C. Schulthess and J.M. MacLaren, *Phys. Rev. B*, **63**, 054416 (2001).
- 26 A. Franco Jr, F.C. e Silva and V.S. Zapf, J. Appl. Phys., 111, 07B530 (2012).
- A. Franco Jr., T.E.P. Alves, E.C. de Oliveira Lima, E. da Silva Nunes and V. Zapf, *Appl. Phys. A*, **94**, 131 (2009).
- P.P. Hankare, S.D. Jadhav, U.B. Sankpal, R.P. Patil, R. Sasikala and I.S. Mulla, *J. Alloys Comp.*, 488, 270 (2009).
- J.Y. Patil, M.S. Khandekar, I.S. Mulla and S.S. Suryavanshi, *Curr. Appl. Phys.*, 12, 319 (2012).
- V. Sepelak, D. Schultze, F. Krumeich, U. Steinike and K.D. Becker, Solid State Ion., 141-142, 677 (2001).
- 31. N.M. Deraz and A. Alarifi, J. Anal. Appl. Pyrolysis, 97, 55 (2012).
- A. Pradeep, P. Priyadharsini and G. Chandrasekaran, J. Magn. Magn. Mater., 320, 2774 (2008).
- T.K. Pathak, N.H. Vasoya, V.K. Lakhani and K.B. Modi, *Ceram. Int.*, 36, 275 (2010).