

Synthesis of Mesoporous Ce1-xSmxO2 Solid Solution Materials and its Catalytic Performance†

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Novel mesoporous structural $Ce_{1-x}Sm_xO_2$ solid solution materials have been synthesized *via* a facile approach of evaporation-induced selfassembly (EISA) in the presence of nitrate as inorganic species and triblock copolymers P_{123} as template. The detailed charaterization by power X-ray diffraction, transmission electron microscope, selected electron diffraction revealed the samples possessing fluorite-structure with the average grain size of 7 nm. Furthermore, N₂ adsorption-desorption measurement illustrated the particular hierarchical porous textural $Ce_{1-x}Sm_xO_2$ solid solution with the pores size in diameter between 2 to 4 nm and the maximun surface area of 106 m²/g. The temperature-programmed reduction of H₂ (H₂-TPR) result confirmed that the ceria-samarium solid solution owned more vacancies than pure CeO₂, hence the double level pores textural $Ce_{1-x}Sm_xO_2$ seem to be potential candidates for catalytic aspect.

Keywords: Ce_{1-x}Sm_xO₂, Solid solution, Mesoporous, Catalytic performance.

INTRODUCTION

Since the ordered mesoporous silica M41S series first successfully synthesized by the scientists of Mobil¹⁻³, the ever increasing number of researchers have devoted themselves to developing advanced meso-textured inorganic or organicinorganic hybrid materials using various types of organic template. Because of well-defined pore structure, uniform pores size in diameter between 2 to 50 nm and large surface area, mesoporous materials were known to apply in the fields of catalysis, optics, photonics, sensors, separation, drug delivery, sorption, acoustic or electrical insulation⁴⁻⁸. However, the conventional amorphous pore wall of most mesoporous materials using surfactant template, limited their application in many fields because of the poor thermal and mechanical stability⁹. Recently, non-silica oxide mesoporous materials, such as transition metal oxides, have been investigated⁴. The transition metal oxides mesoporous materials synthesized by evaporation-induced self-assembly (EISA) method were highly crystallized and possess hierarchical ordered structure on meso-scale as well as on atom-scale9.

Ceria belonging to an indispensable family of transition metal oxides, which is widely used in polishing agents, oxygen sensor, solid oxide fuel cells (SOFC), ceramic materials, espe-

cially catalysis¹⁰⁻¹⁵. Growing number of recent reports have been dedicated to ceria-containing solid solution or composite oxides, which could improve physical/chemical properties of the ceria-based materials, such as catalytic performance. Novel nanocrystalline $Ce_{1-x}La_xO_{2-\delta}(x=0.2)$ solid solution has been successfully prepared by Benjaram, resulting that the $Ce_{1-x}La_xO_{2-\delta}$ (x = 0.2) solid solution exhibited better catalytic properties due to smaller crystallite size, facile reduction, profound bulk oxygen mobility and enhanced OSC¹⁶. Eva Díaz and cooperator researched ceria-zirconia mixed oxides as catalysts for combustion of volatile organic compounds using inverse gas chromatography¹⁷. However, compared to MO_x- CeO_2 (M = Cu, Mn, Fe, Co and Ni) mixed oxides, Sm^{3+} doped CeO₂ solid solution has rarely been reported and thus the unique performance of Ce1-xSmxO2 remains to be further investigated in the experimental process.

The grain size, morphology, pore shape and distribution, oxygen vacancies, which enormously determine the physicochemical properties of materials. In the present work, we report the fabrication of the micro-mesoporous textural $Ce_{1-x}Sm_xO_2$ solid solution *via* evaporation-induced self-assembly by template, meanwhile, the catalytic performance and formation of two series of pores were also investigated.

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EXPERIMENTAL

In this work, 1 g of P_{123} (EO₂₀PO₇₀EO₂₀, Sigma-Aldrich) was dissolved into a mixed solution of 15 mL ethanol and 5 mL deionized water, then a total quantity of 0.005 mol of cerium nitrate [Ce(NO₃)₃·6H₂O,SCRC] and samarium trinitrate [Sm(NO₃)₃, SCRC] on the basis of proportion of Sm³⁺ molar fraction ranging from 0 to 0.2 were added into above mixed solution. The mixture was magnetic stirred for 5 h to ensure homogeneous mixing before being transferred into a 100 mm in diameter petri dishes with a micropipette. Then the petri dishes were placed in an oven for 72 h at 80 °C, after 72 h aging and dry, subsequently the gel was calcined at 400°C for 3 h with a muffle furnace to remove the triblock copolymer P₁₂₃.

The powder X-ray diffraction (PXRD) patterns were observed by using a Rigaku X-ray diffractometer CuK_{α} radiation. The average grain size D was calculated according to Scherrer equation: $D = 0.89\lambda/\beta \cos \theta$, where θ was the diffraction angle of the main peak and β is the full width at half maximum (FWHM). TEM characterization was implemented using a JEM-2010 emission transmission electron microscope operating at an accelerating voltage of 200 kV. The N₂ adsorption-desorption measurement was carried out using Micromeritics ASAP 2010 at 77 K; the specific surface area and the pore size distribution were calculated using Brunauer-Emmett-Teller (BET) and Barrett-Joyner-Halenda (BJH) methods. Reducibility of the samples were examined by H₂-TPR , which were performed with a quartz reactor equipped with a Tianjin XQ TP5000 autoadsorption apparatus.

RESULTS AND DISCUSSION

The XRD patterns of as-synthesized products are shown in Fig. 1. It can be found in this pattern, the typical diffraction peaks at $2\theta = 28.6, 33.1, 47.4, 56.3, 69.5, 76.6$ show the facecentered cubic fluorite structure of ceria (JCPDS No. 34-0394) and exhibit in consistent with the fluorite cubic phase (111), (200), (220), (311), (400) and (331) diffraction planes, respectively. The XRD patterns of ceria-samarium solid solution exhibited shifts of diffraction peaks toward lower 2θ values with respect to pure ceria and can be found becoming broader and lower in intensity with increasing of Sm³⁺ components, which attribute to incorporation of doping of CeO₂ with Sm³⁺. The ionic radius of Sm^{3+} (0.096 nm) is smaller than Ce^{4+} (0.097 nm), hence lattice contraction is expected owing to the partial substitution of Ce⁴⁺ with Sm³⁺. Compared with pure ceria (5.410 Å), the cell parameter "a" of Ce_{1-x}Sm_xO₂ solid solution calculated using the most prominent (111) planes is decreased to about 5.391 Å and is in good agreement with above analysis (Table-1).

TABLE-1 AVERAGE CRYSTALLITE SIZE AND LATTICE PARAMETER OF SAMPLES				
Sample	а	b	с	
Crystallite size (nm)	7.9	7.9	7.7	
Lattice parameter (Å)	5.410	5.394	5.391	

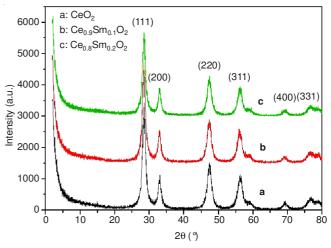


Fig. 1. Power X-ray diffraction patterns of mesoporous Ce1-xSmxO2

Transmission electron microscopic (TEM) and selected electron diffraction (SAED) analyses are undertaken to monitor the evolution of the morphology and structure and to confirm the pore size. As can be seen from the TEM images (Fig. 2a), there are the worm-like porous structure with the size of approximately 2-4 nm, which indicating that the pure CeO₂ by using P₁₂₃ template has mesoporous structure and is made of many small crystals. Fig. 2b illustrates the TEM image of Ce_{0.9}Sm_{0.1}O₂ solid solution. The shape of pores is regularly hexagonal, which are consistent with six square accumulation structure gathered and formed by triblock copolymers P₁₂₃ in the solution. The distances between the adjacent lattice fringes correspond to the interplanar distances of the cubic $CeO_2(111)$, which is $d_{111} = 0.313$ nm (Fig. 2c). TEM image (Fig. 2d) of the Ce_{0.8}Sm_{0.2}O₂ already indicates that the material has small crystals and mesoporous structure, which was due to sintering.

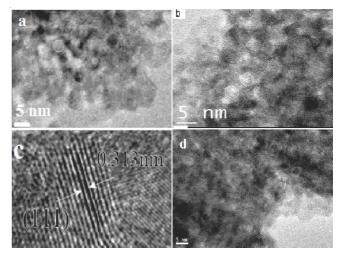


Fig. 2. TEM images of mesoporous CeO_2 (a), $Ce_{0.9}Sm_{0.1}O_2$ (b) $Ce_{0.8}Sm_{0.2}O_2$ (d) and HRTEM images of $Ce_{0.9}Sm_{0.1}O_2$ (c)

The N_2 adsorption-desorption isothrem and pore-size distribution isothern provide further information about structure of samples in Fig. 3. It shows that the curves of the N_2 adsorption-desorption isotherms for three samples are similar and intermediate between types I and IV according to the IUPAC classification^{18,19}.

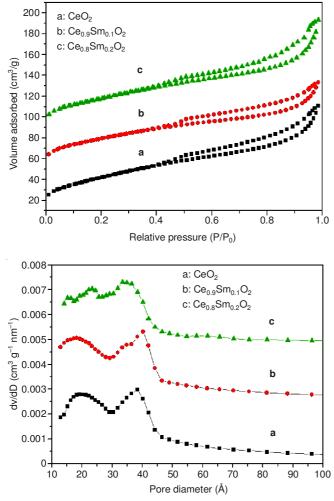


Fig. 3. N₂ adsorption-desorption isotherm and pore-size distribution plot of samples

Its shape indicates the existence of mesopores²⁰, which conforms the results of TEM. The surface area of samples calculated *via* Brunauer-Emment-Teller (BET) are shown in Table-2. The pore-size distribution isotherm suggested that pores size in diameter at 2 to 4 nm, respectively, belonging to the micro- and mesoporous scope. This further confirms the above information on the structure of samples.

TABLE-2					
SURFACE AREA OF SAMPLES					
Sample	а	b	с		
$S_{BET}(m^2 g^{-1})$	110	93	113		

The redox properties of samples are investigated by H₂-TPR experiments. Fig. 4 shows the H₂-TPR of samples synthesized by triblock copolymers as template. The TPR of pure CeO₂ indicated two reduction peaks at 450 and 760 °C. The peak below 500 °C is attributed to the reduction of surface oxygen and peak observed above 700 °C is assigned to the reduction of bulk oxygen²¹. The surface oxygen has higher activity, so it can be reduced under the low temperature. Compared to surface oxygen, the bulk oxygen with low-activity needs to be reduced at a higher temperature. With the introduction of Sm³⁺, H₂-TPR curve of Ce_{1-x}Sm_xO₂ (x = 0.1, 0.2) solid solution can be influenced, the low temperature

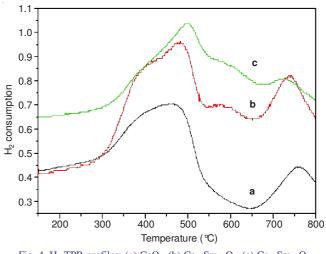


Fig. 4. H₂-TPR profiles: (a) CeO_2 , (b) $Ce_{0.9}Sm_{0.1}O_2$, (c) $Ce_{0.8}Sm_{0.2}O_2$

reduction peak toward higher temperature and the high temperature reduction peak toward lower temperature. The result could be attributed to oxygen vacancy created by the partial substitution of Ce⁴⁺ with Sm³⁺ and the oxygen mobility enhanced *via* mesoporous structure.

Conclusion

The Ce_{1-x}Sm_xO₂(x = 0, 0.1, 0.2) solid solution with mesoporous structure, large surface area and superior catalytic performance have been synthesized *via* EISA method. The Ce_{1-x}Sm_xO₂(x = 0.1, 0.2) solid solution with template structure has a higher oxygen capacity at low-temperature than pure CeO₂, which attribute to its mesoporous structure. The porous structure increases the surface area, which can be proved by the N₂ adsorption and desorption. The higher surface area of Ce_{1-x}Sm_xO₂ provides more surface oxygen, which in agreement with results of predecessors. Moreover, the pore is conducive to mass transfer, so it is good to reducing gas (hydrogen) to the oxide surface, and by reacting to surface oxygen to promote low-temperature reduction process more effectively.

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REFERENCES

- J.S. Beck, J.C. Vartuli, W.J. Roth, M.E. Leonowicz, C.T. Kresge, K.D. Schmitt, C.T.-W. Chu, D.H. Olson and E.W. Sheppard, *J. Am. Chem. Soc.*, 114, 10834 (1992).
- C.T. Kresge, M.E. Leonowicz, W.J. Roth, J.C. Vartuli and J.S. Beck, *Nature*, 359, 710 (1992).

- 3. M.E. Davis, Nature, 417, 813 (2002).
- X.B. Zhao, F. Chen, J. You, X.Z. Li, X.W. Lu and Z.G. Chen, J. Mater. Sci., 45, 3563 (2010).
- X.Z. Li, C.Y. Ni, F. Chen, X. Lu and Z. Chen, J. Solid State Chem., 182, 2185 (2009).
- 6. J. Fan, S. Boettcher and G.D. Stucky, Chem. Mater., 18, 6391 (2006).
- 7. Y. Wan, H.F. Yang and D.Y. Zhao, Acc. Chem. Res., **39**, 423 (2006).
- A.I.Y. Tok, S.W. Du, F.Y.C. Boey and W.K. Chong, *Mater. Sci. Eng. A*, 466, 223 (2007).
- W.H. Shen, X.P. Dong, Y.F. Zhu, H. Chen and J. Shi, *Micropor. Mesopor. Mater.*, 85, 157 (2005).
- S. Masson, P. Holliman, M. Kalaji and P. Kluson, *Mater. Chem.*, 19, 3517 (2009).
- 11. J.L.M. Rupp, A. Infortuna and L. Gauckler, *Acta Mater.*, **54**, 1721 (2006).
- 12. X. Liu, K. Zhou, L. Wang, B. Wang and Y. Li, J. Am. Chem. Soc., 131, 3140 (2009).

- 13. Q.Q. Huang, X.M. Xue and R.X. Zhou, J. Mol. Catal. Chem., 331, 130 (2010).
- 14. P. Borker and A.V. Salker, Mater. Chem. Phys., 103, 366 (2007).
- W. Zhang, D. Zhang, T.X. Fan, J.J. Gu, J. Ding, H. Wang, Q.X. Guo and H. Ogawa, *Chem. Mater.*, **21**, 33 (2009).
- 16. B.M. Reddy, L. Katta and G. Thrimurthulu, *Chem. Mater.*, **22**, 467 (2010).
- 17. E. Díaz, B. de Rivas, R. López-Fonseca, S. Ordóñez and J.I. Gutiérrez-Ortiz, J. Chromatogr. A, **1116**, 230 (2006).
- 18. H.Q. Zhu, Z.F. Qin, W.J. Shan, W.J. Shen and J.G. Wang, *Catal*, **225**, 267 (2004).
- 19. K.S. Sing, D.H. Everett and R.A.W. Haul, Appl. Chem, 57, 603 (1985).
- V. Perrichon, A. Laachir, S. Abouarnadasse, O. Touret and G. Blanchard, Appl. Catal. A, 129, 69 (1995).
- 21. J.M. Zhou, L. Zhao, Q.Q. Huang, R.X. Zhou and X.K. Li, *Catal. Lett.*, **127**, 277 (2009).