

# Synthesis and Studies on Electroactive Poly(3-chloroaniline), Poly(3-chloroaniline)-Blend-polyethyleneglycol and Poly(3-chloroaniline)-Blend-Poly(vinyl alcohol)

L. PORSELVI and S. JHANCY MARY\*

PG and Research Department of Chemistry, Auxilium College, Vellore-632 006, India

\*Corresponding author: Fax: +91 416 2 2247281; E-mail: jhancy2011@gmail.com

Received: 25 March 2013;

Accepted: 8 August 2013; Published online: 30 January 2014;

AJC-14637

The poly(3-chloroaniline), poly(3-chloroaniline)-blend-polyethyleneglycol and poly(3-chloroaniline)-blend-poly(vinyl alcohol) were prepared by *in situ* oxidative polymerization technique using ammonium per sulphate as oxidizing agent and HCl as dopant. The polymer and its blends were characterized by FT-IR, UV-visible spectroscopy, thermogravimetric analysis and electrical conductivity. The synthesized polymer and its blends show the presence of benzenoid structure, quinonoid structure and the formation of the charged species (Q=N<sup>+</sup> H-B or B-N<sup>+</sup> H-B). Absorption band appear in the range 450-550 nm in polyaniline, whereas in the poly(3-chloroaniline) and its blends it is shifted to lower wave length because of the electronic effect which has reduced the amount of delocalization there by increasing the energy gap. The complex impedance of the polymer and its blends were measured by using four point probe method. The electrical conductivities of poly(3-chloroaniline), poly(3-chloroaniline)-blend-polyethyleneglycol and poly(3-chloroaniline)-blend-poly(vinyl alcohol) were  $1.37 \times 10^{-11}$  S cm<sup>-1</sup>,  $3.67 \times 10^{-11}$  S cm<sup>-1</sup> and  $2.64 \times 10^{-9}$  S cm<sup>-1</sup> respectively. The blends synthesized were of semiconducting nature and poly(3-chloroaniline)-blend-poly(vinyl alcohol) has higher conductivity than poly(3-chloroaniline) and poly(3-chloroaniline)-blendpolyethyleneglycol.

Keywords: Poly(3-chloroaniline), Poly(vinyl alcohol), Polyethyleneglycol, Four point probe method, Complex impedence.

### **INTRODUCTION**

Electrically conducting polymers have raised a great deal of interest after the successful synthesis of conducting polyacetylenes by Shirakawa *et al.*<sup>1</sup>. Conducting polymers also known as synthetic metals are polymers with a highly  $\pi$ -conjugated polymeric chain<sup>2-4</sup>. Typically conducting polymers include polyacetylene, polyaniline, polypyrrole, polythiophene, poly(*para*-phenylene), polyphenylenevinylene, polyfuran, *etc.* These conjugated polymers can be electrical insulators, semiconductors or conductors, depending on the level of doping and nature of the dopants.

Among these conductive polymers, polyaniline has special advantages, such as ease of synthesis, environmental stability, low cost of the monomer, nonredox doping by protonic acids and ready modification of the oxidation states of the polymer chain<sup>5</sup>. The applications of polyaniline in various important fields include active electrode<sup>6,7</sup>, electromagnetic shielding materials, microelectronic materials and electrochromic device, metal anti-corrosive coating<sup>8,9</sup>, anti-static coating<sup>10</sup>, rechargeable batteries<sup>11,12</sup>, energy storage and transfer, redox micro-template<sup>13</sup>, indicators and sensors<sup>14,15</sup>. The conducting polyaniline microparticles can be electrostatically

accelerated to hypervelocities, indicating their suitability as mimics of solar system dusts for the calibration of impact ionization detectors for space craft<sup>16</sup>. Polyaniline also shows high gas-separation ability with the highest ideal oxygen/ nitrogen separation factor of up to 30<sup>17</sup>.

However, the applications of polyaniline have been limited by its intractable nature as it is usually obtained chemically as an insoluble powder and electrochemically as a thin brittle film<sup>18</sup>. Considerable progress has been made in the last few years in the processibility of conducting polymers by the synthesis of polymer blends and composites<sup>19,20</sup> and soluble polyaniline derivatives and copolymers<sup>21,22</sup>. The processability and solubility of polyaniline were improved by using various groups at the ring or N-substituents. Electron donating groups such as alkyl, alkoxy and amino, or electron withdrawing groups such as carboxyl, sulpho and cyano decrease orbital overlap of the  $\pi$ -electrons and the nitrogen lone pair that may increase the solubility and readiness for processability. However, they have been found to be more or less conducting than polyaniline depending on the nature of the blending materials, even though they are more soluble $^{23}$ .

The synthesis of different homopolymers and copolymers derived from anilines with electron with drawing groups has

been reported. Ranger and Leclrc<sup>24</sup> synthesized poly (cyanoanilines) and poly(cyanoaniline-*co*-aniline) by chemical methods. Yu *et al.*<sup>25</sup> reported the chemical copolymerizations of aniline with mono-*ortho*-halogenated anilines (2-chloroaniline and 2-iodoaniline). The synthesis of poly(aniline-*co*-2-chloroaniline)-*p*-toluene-sulfonate and poly(aniline-*co*-2-chloroaniline) formate salts by chemical polymerization has been reported by Palaniappan<sup>26</sup>. A few studies on the oxidative polymerization of mono- and di-substituted anilines with electron with drawing groups have been reported<sup>25,27</sup>.

An attempt has been made to chemically synthesize 3-chloro substituted polyaniline and its blends using *in situ* oxidative polymerization technique and to study their thermal and electrical properties. A study on the effect of the substituent on the rate of polymerization, thermal stability and electrical conductivity may also be made. The attempt to synthesize blends of poly(3-chloroaniline) could open up avenues in the field of research in conducting polymers with interesting commercial applications.

#### **EXPERIMENTAL**

Poly(3-chloroaniline) and its blends were chemically synthesized in aqueous acidic medium using ammonium per sulphate  $(NH_4)_2S_2O_8$  as the oxidizing agent using modifications in the method already reported in the literature<sup>28</sup>.

Polymerization of 3-chloroaniline (monomer: oxidant = 1:1): 10.9 mL of 3-chloroaniline (0.0858 mol) was mixed with 50 mL of water in the reaction vessel. 19.6 g of ammonium persulphate (0.0858 mol) was dissolved in 50 mL water. 20 mL of concentrated hydrochloric acid was taken separately in a 100 mL beaker. The reaction vessel containing 3-chloroaniline was placed in the freezing mixture. Added the oxidant and acid alternatively in drops with constant stirring using the magnetic stirrer over a period of 2 h. After the addition was over, the stirring was continued for 3 h. The product formed was kept in the refrigerator overnight and then filtered. The solid product formed was again transferred into a 250 mL beaker, suspended in water and stirred for 1 h on a magnetic stirrer, washed several times with water and filtered. The washing of the product was continued till the filtrate was free of acid. The pale brown product formed was dried and finely powdered. The synthesis using the monomer oxidant ratio 1:1 resulted in low yield.

**Polymerization of 3-chloro aniline (monomer: oxidant** = 1:2): 5.45 mL of 3-chloro aniline (0.0427 mol) was mixed with 25 mL of water in the reaction vessel. 19.6 g of ammonium persulphate (0.0858 mol) was dissolved in 50 mL water. 20 mL of concentrated hydrochloric acid was taken separately in a 100 mL beaker. The reaction vessel containing 3-chloro aniline was placed in the freezing mixture. The oxidant and the acid were also kept in the freezing mixture. Added the oxidant and acid alternatively in drops with constant stirring using the magnetic stirrer over a period of 2 h. After the addition was over, the stirring was continued for 3 h. The product formed was kept in the refrigerator overnight and then filtered.

The product formed was again transferred into a 250 mL beaker, suspended in water and stirred for 1 h on a magnetic stirrer, washed several times with water and filtered. The washing of the product was continued till the filtrate was free of acid. The product formed was dried and finely powdered. A dark grey solid was obtained. The yield of the product was 3 g.

The synthesis of the polymer when the monomer: oxidant ratio 1:1 was not successful as confirmed by FTIR spectral studies. Only the synthesis when the monomer: oxidant ratio was 1:2 resulted in polymerization. Hence the poly(3-chloro aniline)-blend-polyethyleneglycol and poly(3-chloroaniline)blend-poly(vinyl alcohol) were also synthesized by using the monomer: oxidant ratio as 1:2.

Synthesis of poly(3-chloroaniline)-blend-polyethy leneglycol and poly(3-chloroaniline)-blend-poly(vinyl alcohol): The method adopted was polymerization of a monomer in a polymer matrix. 5 g of polyethyleneglycol/ poly(vinyl alcohol) was dissolved in 100 mL of water and added 50 mL of concentrated hydrochloric acid in the reaction vessel. 3-Chloroaniline (5.5 mL) was mixed with 7.2 mL of concentrated hydrochloric acid and 25 mL of water. 19.6 g of ammonium persulphate was dissolved in 46 mL of water. The beaker containing polyethyleneglycol/poly(vinyl alcohol) in concentrated hydrochloric acid was kept in the freezing mixture. The monomer suspended in water, hydrochloric acid and the oxidant solution were added drop wise to it with constant stirring over a period of 1.5 h.

The stirring was continued for 3 h. The product was placed in the refrigerator overnight and filtered. The blend was again transferred into a 250 mL beaker, suspended in water and stirred for 1 h on a magnetic stirrer, washed with water and filtered. The washing of the products were continued till the filtrates were free of acid. The products formed were dried and powdered. The poly(3-chloroaniline), poly(3-chloroaniline)-blend-polyethyleneglycol and poly(3-chloroaniline)blend-poly(vinyl alcohol) were dark grey, dark blue and dark green respectively and they were soluble in DMSO and DMF.

In the present investigation, transmittance spectra were used to quantify the characteristic peaks of the polymer and its blends in the mid infrared region between 4000 to 400 cm<sup>-1</sup> with 0.5 cm<sup>-1</sup> resolution, using Thermo Nicolet-Model:Magna 550. The UV-visible spectra of the polymer and its blends in DMSO were recorded from 200-800 nm wavelength using a Systronics double beam UV-visible spectrophotometer 2201. The thermo gravimetric analysis of the poly(3-chloroaniline) and its blends were carried out in a Q50 V20.6 Build 31 thermal analyzer from 27 °C to 700 °C at a heating rate of 15 °C min<sup>-1</sup> under nitrogen atmosphere with gas flow rate of 90 mL min<sup>-1</sup>.

The complex impedance measurements were carried out on pelleted specimens of poly(3-chloroaniline) and its blends using a Hewlett Packard model HP4284A precision LCR meter in the frequency range 20 HZ - 1 MHZ and in the temperature range 295-445 K. Accordingly, the synthesized solid polymer and its blends were ground into fine powders and disc shaped pellets of 13 mm diameter and the required thickness were prepared by pressing the powder samples in an IR sample press under a pelletizing pressure of 7 ton/cm<sup>2</sup> to form circular pellets. The thicknesses of the pellets were measured using a screw gauge.

## **RESULTS AND DISCUSSION**

The FTIR spectrum of poly(3-chloroaniline) as in Fig. 1 shows the presence of both benzenoid and quinonoid rings. A strong peak at 1585 cm<sup>-1</sup> confirms the presence of quinonoid ring and that at 1508 cm<sup>-1</sup> is characteristic of a benzenoid structure<sup>29,30</sup>. N-H stretching frequencies are observed at 3392 cm<sup>-1</sup> in the polymer and a doublet at 3500 cm<sup>-1</sup> in the monomer. A comparison of the spectra with the IR spectrum of the monomer indicates an increase in symmetry. The stretching and bending frequencies are tabulated in Table-1 and assignments have been made based on literature. According to studies made by Tang et al.<sup>29</sup>, the presence of a band around 1145 cm<sup>-1</sup> can be related to the vibrational mode of protonated amines generated during the acid doping process. This peak appears around 1145 cm<sup>-1</sup> in the FTIR spectrum of poly(3chloroaniline) due to (Q=N<sup>+</sup> H-B or B-N<sup>+</sup> H-B). The band at 1164 cm<sup>-1</sup> corresponds to the bending of C-H in plane mode of N=Q=N as reported. The C-Cl stretching frequency is observed around 792 cm<sup>-1</sup>.

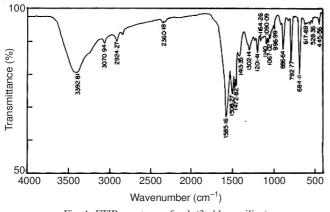


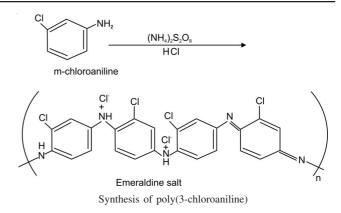
Fig. 1. FTIR spectrum of poly(3-chloroaniline)

TABLE-1 ASSIGNMENTS OF THE MAIN PEAKS IN THE FTIR SPECTRUM OF POLY(3-CHLOROANILINE)

Peak assignment	Wave number (cm <sup>-1</sup> )	
N- H Stretching vibration	3392	
C- H Stretching vibration of aromatic ring	3070	
-C=C- stretching of quinonoid	1585	
-C=C- stretching of benzenoid	1508	
Stretching of C- N aromatic	1302	
Bending of -C – H in plane mode of N=Q=N	1164	
C- Cl stretching vibration	792	
B- N <sup>+</sup> H- B / Q=N <sup>+</sup> H- B	1140	
Structure of N=Q=N	1201	

Analogues to the studies made on polyaniline and based on the interpretation of the FTIR data, the following reaction has been proposed.

Fig. 2. shows the absorption spectrum of poly(3-chloroaniline). The spectrum of 3-chloroaniline in DMF exhibits a band at 216 nm. In poly(3-chloroaniline) it is shifted to a higher wavelength of 246 nm. An absorption band at 246 nm is due to the  $\pi$ - $\pi^*$  transition of the benzenoid and quinonoid rings. The bathochromic shift indicates an increase in the extent of delocalization. In the case of polyaniline, absorption bands



appear in the range 450-550 nm, which is the characteristic absorption spectrum of the salt form of polyaniline<sup>31-33</sup>.  $\pi$ -Electrons move from one ring to another and the  $\pi$ -electron delocalization takes place over the entire molecule. The greater the amount of delocalization, the smaller the energy gap between the  $\pi$ -orbitals. In the case of poly(3-chloroaniline), the steric factor and the electronic effect reduce the amount of delocalization there by increasing the energy gap and so the transition occurs at lower wavelength, lower than poly aniline. The substituent in the *meta* position to the amino group reduces the electron density to a greater extent. The weak absorption at 502 nm is due to  $n-\pi^*$  transition.

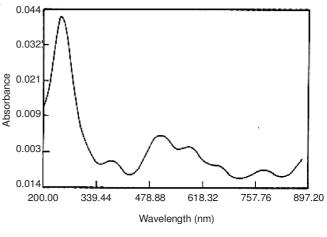


Fig. 2. UV-visible spectrum of poly(3-chloroaniline)

DTA-TGA of poly(3-chloroaniline) is given in Fig. 3. The decomposition of the polymer occurs prominently and exothermally at a peak temperature of 261 °C. Water is lost at 100 °C which is immediately followed by the exothermic decomposition of the polymer. It is seen that the exothermicity is high as evidenced by a broad peak. This might have compensated the loss of water. At this temperature the weight loss is about 60.75 %. It is also noted that even after 600 °C, complete decomposition has not taken place. The presence of chlorine atom has prevented the complete decomposition of the pure polymer.

The FTIR spectrum of poly(3-chloroaniline)-blendpolyethyleneglycol is shown in Fig. 4. Assignments of the main peaks in the FTIR spectrum of the blend are presented in Table-2. The N-H stretching frequency appears at 3408 cm<sup>-1</sup>.

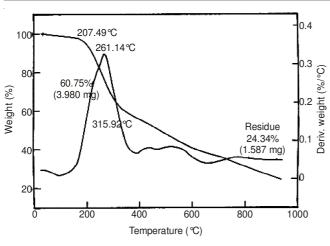


Fig. 3. DTA / TGA of poly(3-chloroaniline)

Since it is a singlet it shows that *m*-chloro aniline has undergone polymerization and there is no free NH<sub>2</sub> group. The peak at 1596 cm<sup>-1</sup> is due to the quinonoid ring and the peak at 1378 cm<sup>-1</sup> is due to the benzenoid ring<sup>29,30</sup> structure. The C-Cl band occurs at 778 cm<sup>-1</sup>. The OH band appears at 3829 cm<sup>-1</sup>. Since the IR spectrum contains all the main bands of poly(3chloroaniline) and polyethyleneglycol, it shows that blending has taken place. The peak at 1114 cm<sup>-1</sup> is due to the emeraldine salt structure which contributes to the electrical conductivity of the blend as reported by Tang *et al.*<sup>29</sup>. The intensity of the bands has decreased which indicates that the polymerization has taken place.

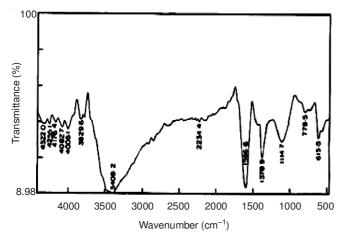


Fig. 4. FTIR spectrum of poly(3-chloroaniline)-blend-polyethylene glycol

TABLE-2		
ASSIGNMENTS OF THE MAIN PEAKS IN THE FTIR		
SPECTRUM OF POLY(3-CHLOROANILINE)-BLEND		
POLYETHYLENEG	LYCOL	
ak assignment	Wave number (cm <sup>-1</sup>	
H stretching frequency	3408	
=C- stretching of aninonoid /		

Pea

N-

-C=C- stretching of quinonoid /	1596
N-H bending vibration	1590
-C=C- stretching of benzenoid	1378
$B-N^+H-B/Q=N^+H-B$	1114
-C-Cl stretching vibration	778
-CH-OH polyvalent alcohols	3829

Fig. 5. shows the optical absorption spectrum of poly(3chloroaniline)-blend-polyethyleneglycol. The spectrum of the blend in DMSO exhibits band at 225 nm and 293 nm. The absorption band at 225 nm is due to  $\pi$ - $\pi$ \* transition of the benzenoid and quinonoid rings and the band 293 nm is due to n- $\pi$ \* transition. It occurs at longer wavelength. When compared to polyaniline, it is shifted to lower wavelength because of the electronic effect which has reduced the amount of delocalization there by increasing the energy gap.

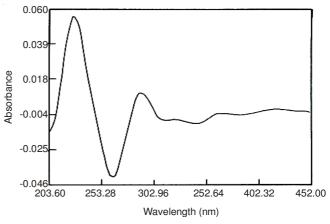


Fig. 5. UV-visible spectrum of poly(3-chloroaniline)-blend-poly ethyleneglycol

The TGA of the blend of poly(3-chloroaniline)-blendpolyethyleneglycol is given in Fig. 6. The sample decomposes in two steps. Initially there is loss of water and there is loss of the HCl at 249 °C. Then the decomposition of the polymer is continuous upto 692 °C and even after 690 °C the complete decomposition has not taken place. The decomposition of the blend leaves a char content of 21.54 %. This shows that the presence of chlorine has prevented the complete decomposition of the blend.

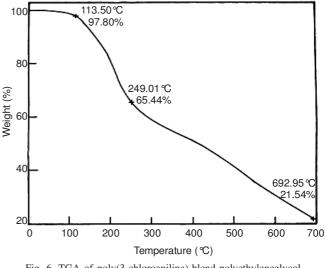


Fig. 6. TGA of poly(3-chloroaniline)-blend-polyethyleneglycol

The FTIR spectrum of poly(3-chloroaniline)-blendpoly(vinyl alcohol) is shown in Fig. 7. Assignments of the main peaks in the FTIR spectrum of the blend are presented in Table-3. The N-H stretching frequency appears at 3399 cm<sup>-1</sup>. Since it is singlet it shows that 3-chloroaniline has undergone polymerization and there is no free NH<sub>2</sub> group. The peak at 1596 cm<sup>-1</sup> is due to the quinonoid ring and the peak at 1381 cm<sup>-1</sup> is due to the benzenoid ring<sup>29,30</sup> structure. The -C-Cl band occurs at 769 cm<sup>-1</sup>. The -OH bands appear at 3786 cm<sup>-1</sup>. Since the IR spectrum contains all the main bands of poly(3-chloroaniline) and poly(vinyl alcohol), it shows that blending taken place. The peak at 1124 cm<sup>-1</sup> is due to the emeraldine salt structure which contributes to the electrical conductivity of the blend as reported by Tang *et al.*<sup>29</sup>. The intensity of the bands has decreased which indicates that the polymerization has taken place.

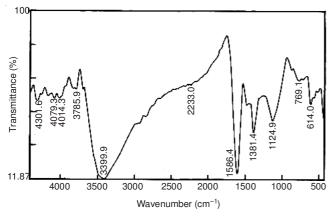


Fig. 7. FTIR spectrum of poly(3-chloroaniline)-blend-poly(vinyl alcohol)

TABLE-3 ASSIGNMENTS OF THE MAIN PEAKS IN THE FTIR SPECTRUM OF POLY(3-CHLOROANILINE)-BLEND-POLYVINYLALCOHOL

Peak assignment	Wavenumber (cm <sup>-1</sup> )
N-H stretching vibration	3399
-C=C- stretching of quinonoid / N-H bending vibration	1596
-C=C- stretching of benzenoid	1381
$B-N^{+}H-B/Q=N^{+}H-B$	1124
-C-Cl stretching vibration	769
-CH-OH (polyvalent alcohols)	3786

Fig. 8 shows the optical absorption spectrum of poly(3chloroaniline) with poly(vinyl alcohol). The spectrum of the

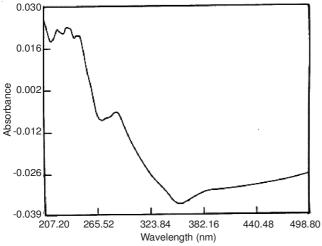


Fig.8. UV-visible spectrum of poly(3-chloroaniline)-blend-poly(vinyl alcohol)

blend in DMSO exhibits band at 207 nm and 243 nm. The absorption band at 207 nm is due to  $\pi$ - $\pi^*$  transition of the benzenoid and quinonoid rings and the band 245 nm is due to n- $\pi^*$  transition. It occurs at longer wave length. When compared to polyaniline it is shifted to lower wave length because of the electronic effect which has reduced the amount of delocalization there by increasing the energy gap.

The TGA of the blend of poly(3-chloroaniline) and poly(vinyl alcohol) is given in Fig. 9. The sample decomposes in two steps. The initial loss of water takes place and then the sample decomposes continuously upto 535 °C. Even after 690 °C, 48.7 % of the residue remains. This shows that the presence of chlorine has prevented the complete decomposition of the blend.

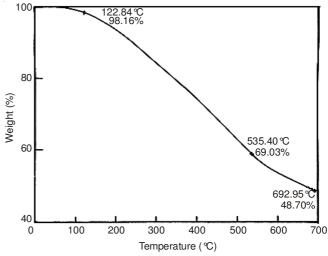


Fig. 9. TGA of blend of poly(3-chloroaniline)-blend-poly(vinyl alcohol)

**Electrical conductivity studies:** The polymer and its blends were finely powdered and pressed into a pellet using IR Sample Press using Model AVAPAR-330 Thermonicolate at a maximum pressure of 15 tons/cm<sup>2</sup> to form a circular pellet of 12 mm diameter. The thickness of the pellet was measured using screw gauge. The measurement was done by placing the pellet between two silver electrodes and the bulk resistance ( $R_b$ ) was measured using computational analysis using HPVEE programmes and the values are tabulated in Tables-4-6.

TABLE-4				
,	COMPLEX IMPEDANCE MEASUREMENT OF POLY(3-CHLORO ANILINE)			
	OFPOLI	(3-CHLORO P	AINILIINE)	
Frequency	Impedance	$\theta$ in degree	$Z \cos \theta$	-Z sin θ
in Hz	in ohm	-		
1000000	14709	-89.28	186	14708
320000	45120	-89.17	651	45115
100000	142847	-89.02	2444	142826
35294	400710	-88.91	7621	400638
8000	1738120	-88.63	41447	1737626
2000	6826300	-88.02	235710	6822229
500	26488900	-87.46	1176214	26462773
100	125655000	-87.82	4771463	125564375
20	731266000	-90.72	-9204432	731208070

The plot gives  $-Z\sin\theta vs$ .  $Z\cos\theta$  as in Figs. 10-12. The point of intersection of the impedance plot on the real axis at

 TABLE-5

 COMPLEX IMPEDANCE MEASUREMENT OF POLY(3 

 CHLOROANILINE)-BLEND-POLYETHYLENEGLYCOL

 Frequency Impedance

 mpedance

 θ in degree
 Z cos θ
 -Z sin θ

in Hz	in ohm	0 III degree	2 005 0	-2. Sin 0
1000000	52510	-83.41	6029	52163
320000	150115	-84.53	14310	149431
100000	451466	-85.04	38997	449779
35294	1211570	-85.12	103152	1207171
8000	4930390	-84.94	434949	4911167
2000	18446600	-84.56	1749889	18363413
500	67473400	-84.11	6918659	67117746
100	3.39E + 08	-96.32	-3.7E + 07	3.37E + 08
20	1.38E + 09	-95.45	-1.3E + 08	1.37E + 09

TABLE-6 COMPLEX IMPEDANCE MEASUREMENT OF POLY(3-CHLOROANILINE)-BLEND-POLYVINYLALCOHOL

Frequency in Hz	Impedance in ohm	$\theta$ in degree	$Z\cos\theta$	-Z sin θ
1000000	63162	-69.68	22009	59204
320000	153779	-70.41	51569	144875
100000	384649	-68.40	141597	357638
35294	841475	-65.24	352449	764107
8000	2371810	-59.91	1189116	2052191
2000	5747700	-55.38	3265538	4729939
500	13488300	-51.28	8437190	10523691
100	33971000	-36.73	27227905	20314282
20	54610700	-15.63	52590052	14717844

the high frequency region within the frequency range under investigation gives the bulk resistance of the sample.

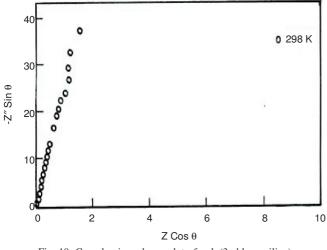


Fig. 10. Complex impedance plot of poly(3-chloroaniline)

The electrical conductivities of the poly(3-chloroaniline), poly(3-chloroaniline)-blend-polyethyleneglycol and poly(3-chloroaniline)-blend-poly(vinyl alcohol) are calculated using the formula.

$$\sigma = \frac{t}{R \times A}$$

where 't' is the thickness of the pellet, 'A' is the area of the cross section and 'R' is the bulk resistance of the individual sample.

The electrical conductivity of the blend of poly(3-chloroaniline) is

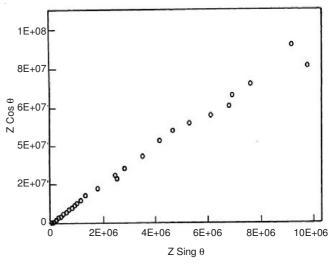


Fig. 11. Complex impedance plot of poly(3-chloroaniline)-blendpolyethyleneglycol

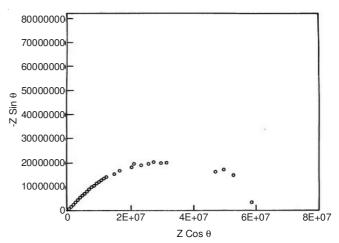


Fig. 12. Complex impedance plot of poly(3-chloroaniline)-blend-poly(vinyl alcohol)

$$\sigma = 0.31 \times 10^{-1}/1.7086 \times 10^{9} \times 1.3279$$
$$= 1.37 \times 10^{-11} \text{ S cm}^{-1}$$

The electrical conductivity of poly(3-chloroaniline)blend-polyethyleneglycol is

$$\sigma = 0.102/2.40 \times 10^9 \times 1.1304$$

$$= 3.67 \times 10^{-11} \text{ S cm}^{-1}$$

The electrical conductivity of poly(3-chloroaniline)blend-poly(vinyl alcohol) is

$$\sigma = 0.2 / 6.4 \times 10^7 \times 1.1304$$

$$= 2.64 \times 10^{-9} \text{ S cm}^{-1}$$

The electrical conductivities of the poly(3-chloroani line), poly(3-chloroaniline)-blend-polyethyleneglycol and poly(3-chloroaniline)-blend-poly(vinyl alcohol) are  $1.37 \times 10^{-11}$  S cm<sup>-1</sup>,  $3.67 \times 10^{-11}$  S cm<sup>-1</sup> and  $2.64 \times 10^{-9}$  S cm<sup>-1</sup>, respectively.

## Conclusion

Poly(3-chloroaniline), poly(3-chloroaniline)-blendpolyethyleneglycol and poly(3-chloroaniline)-blendpoly(vinyl alcohol) have been successfully synthesized, characterized, thermal and electrical properties have been studied. The structure of the polymer and its blends consists of two types of rings, the quinonoid and benzenoid as shown by FTIR spectrum. The presence of B-N<sup>+</sup> H-B/Q = N<sup>+</sup> H-B is supported by the presence of the peaks around 1140 cm<sup>-1</sup>, 1114 cm<sup>-1</sup> and 1124 cm<sup>-1</sup> in the polymer and its blends with polyethyleneglycol and poly(vinyl alcohol) respectively. The  $\lambda_{max}$  values in UV-visible spectra were lower than that of polyaniline due to the presence of chlorine in the meta position which has decreased the  $\pi$ -electron delocalization. The TGA shows that the presence of chlorine has prevented the complete combustion of the blends.

The conductivities of the polymer and its blends as measured by four point probe method were

(1) Poly (*m*-chloro aniline) =  $1.37 \times 10^{-11}$  S cm<sup>-1</sup>

(2) Poly(3-chloroaniline)-blend-polyethyleneglycol =  $3.67 \times 10^{-11}$  S cm<sup>-1</sup>

(3) Poly(3-chloroaniline)-blend-poly(vinyl alcohol) =  $2.64 \times 10^{-9} \text{ S cm}^{-1}$ 

The blends synthesized were of semiconducting nature and poly(3-chloroaniline)-blend-poly(vinyl alcohol) has higher conductivity than poly(3-chloroaniline) and poly(3-chloroaniline)-blend-polyethyleneglycol. This piece of work provides scope for further research in the field of conducting substituted polyaniline blends.

#### REFERENCES

- H. Shirakawa, E.J. Louis, A.G. MacDiarmid, C.K. Chiang and A.J. Heeger, J. Chem. Soc., Chem. Commun., 16, 578 (1977).
- 2. H. Shirakawa, Angew. Chem. Int. Ed., 40, 2575 (2001).
- 3. A.G. MacDiarmid, Angew. Chem. Int. Ed., 40, 2581 (2001).
- 4. A.J. Heeger, Angew. Chem. Int. Ed., 40, 2591 (2001).
- G. Tourrillan and Skotheim TA, Handbook of Conducting Polymers; Marcel Dekker, New York, p. 293 (1986)
- K.S. Ryu, K.M. Kim, N.G. Park, Y.J. Park and S.H. Chang, J. Power Sources, 103, 305 (2002).
- S.K. Dhawan, N. Singh and S. Venkatachalam, *Synth. Met.*, **125**, 389 (2001).
- P. Somani, A.B. Mandale and S. Radhakrishnan, *Acta Mater.*, 48, 2859 (2000).

- 9. K.G. Conroy and C.B. Breslin, Electrochim. Acta., 48, 721 (2003).
- E.P. Maziarz III, S.A. Lorenz, T.P. White and T.D. Wood, J. Am. Soc. Mass. Spectrom., 11, 659 (2000).
- 11. A. Mirmohseni and R. Solhjo, Eur. Polym. J., 39, 21 (2003).
- M.S. Rahmanifar, M.F. Mousavi and M. Shamsipur, J. Power Sources, 110, 229 (2002).
- 13. M.A. Khan, C. Perruchot, S.P. Armes and D.P. Randall, *J. Mater. Chem.*, **11**, 2363 (2001).
- 14. Z. Jin, Y.X. Su and Y.X. Duan, Sens. Actuators B, 71, 118 (2000).
- 15. D. Nicolas-Debarnot and F. Poncin-Epaillard, *Anal. Chim. Acta*, **475**, 1 (2003).
- M.J. Burchell, M. Willis, S.P. Armes, M.A. Khan, M.J. Percy and C. Perruchot, *Planet. Space Sci.*, **50**, 1025 (2002).
- M.R. Anderson, B.R. Mattes, H. Reiss and R.B. Kaner, *Science*, 252, 1412 (1991).
- H.S.O. Chan, S.C. Ng, W.S. Sim, K.L. Tan and B.T.G. Tan, *Macromole*cules, 25, 6029 (1992).
- J. Anand, S. Palaniappan and D.N. Sathyanarayana, *Prog. Polym. Sci.*, 23, 993 (1998).
- A. Pud, N. Ogurtsov, A. Korzhenko and G. Shapoval, *Prog. Polym. Sci.*, 28, 1701(2003).
- W.Y. Zheng, K. Levon, J. Laakso and J.E. Osterholm, *Macromolecules*, 27, 7754 (1994).
- M.T. Nguyen, P. Kasai, J.L. Miller and A.Z. Diaz, *Macromolecules*, 27, 3625 (1994).
- R.N. Mohammad, S.T. Soheil, S. Roya and A.E. Ali, *Iran. Polym. J.*, 17, 1 (2008).
- 24. M. Ranger and M. Leclerc, Synth. Met., 84, 85(1997).
- Y. Sahin, S. Perçin and G.Ö. Alsancak, J. Appl. Polym. Sci., 89, 1652 (2003).
- 26. S. Palaniappan, Eur. Polym. J., 33, 1735 (1997).
- A.L. Sharma, V. Saxena, S. Annapoorni and B.D. Malhotra, J. Appl. Polym. Sci., 81, 1460 (2001).
- 28. U.S. Waware and S.S. Umare, React. Funct. Polym., 65, 343 (2005).
- 29. J. Tang, X. Jing, B. Wang and F. Wang, Synth. Met., 24, 231 (1998).
- Y.H. Kim, C. Foster, J. Chiang and A.J. Heeger, *Synth. Met.*, 25, 49 (1998).
- 31. M. Wan, Synth. Met., 31, 51 (1989).
- 32. H. Kuzmany and N.S. Sariciftci, Synth. Met., 18, 353 (1987).
- A.P. Monkman, D. Bloor, G.C. Stevens and J.C.H. Stevens, J. Phys. D: Appl. Phys., 20, 1337 (1987).