



## Synthesis of Metal Ion Doped Sr<sub>2</sub>CeO<sub>4</sub> by Hydrothermal Method and Its Luminescent Properties

YONG-QING ZHAI\*, ZHI-CHUN HU, JIAN MA, XUAN LI and JIN-HANG LI

\*College of Chemistry and Environmental Science, Hebei University, Baoding 071002, P.R. China

\*Corresponding author: Fax: +86 312 5079525; E-mail: zhaiyongqinghu@163.com

Received: 19 August 2013;

Accepted: 14 November 2013;

Published online: 16 July 2014;

AJC-15568

Blue-white emitting phosphor Sr<sub>2</sub>CeO<sub>4</sub> and a series of Sr<sub>2</sub>CeO<sub>4</sub>:M (M = Mg, Ca, Zn, Al) were synthesized by hydrothermal method using (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub> as the precipitation agent. Phase structure and luminescence properties of the as-synthesized phosphors were investigated by X-ray diffraction and fluorescence spectrophotometry respectively. Results show that Sr<sub>2</sub>CeO<sub>4</sub>:M (M = Mg, Ca, Zn, Al) phosphors have the same orthorhombic crystal structure as that of Sr<sub>2</sub>CeO<sub>4</sub>. Doping with metal ions has little effect on shape and position of peaks in emission spectrum of Sr<sub>2</sub>CeO<sub>4</sub>, which is located in the range of 400-600 nm with main peak at 466 nm and ascribed to the f→l<sub>g</sub> transition of Ce<sup>4+</sup> ions. Among these metal ions, doping with Zn<sup>2+</sup> or Mg<sup>2+</sup> can effectively enhance the photoluminescence of Sr<sub>2</sub>CeO<sub>4</sub> phosphor, the appropriate doping concentration of Zn<sup>2+</sup> and Mg<sup>2+</sup> is 5 and 4 %, respectively. Moreover, a red-shift is found in the excitation spectrum of Sr<sub>2</sub>CeO<sub>4</sub> phosphor after doping with Zn<sup>2+</sup> or Mg<sup>2+</sup>.

**Keywords:** Sr<sub>2</sub>CeO<sub>4</sub>, Metal ion-doping, Hydrothermal synthesis, Photoluminescence.

### INTRODUCTION

It is well known that oxide-based phosphors have been widely utilized in the fields of optoelectronics and display devices due to their high thermal stability and chemical stability<sup>1,2</sup>. Danielson *et al.*<sup>3</sup> have successfully synthesized a unusual rare-earth oxide phosphors Sr<sub>2</sub>CeO<sub>4</sub> by combinatorial technique, emitting a blue-white fluorescence by the excitation through the ultraviolet ray, cathode ray or X-ray. Further research suggests that the emission of Sr<sub>2</sub>CeO<sub>4</sub> is originated from Ce<sup>4+</sup> to O<sup>2-</sup> charge transfer (CT) transition. The broad excitation spectrum in the 200-400 nm range indicated the feasibility of Sr<sub>2</sub>CeO<sub>4</sub> phosphor utilized in UV-LED. Moreover, owing to its broad emission band, Sr<sub>2</sub>CeO<sub>4</sub> can also be used as the matrix materials. Until now, the rare earth-doped Sr<sub>2</sub>CeO<sub>4</sub> (Sr<sub>2</sub>CeO<sub>4</sub>: Ln = Eu, Dy, Sm, Ho, Er, Tm) has been widely investigated as a kind of fluorescent materials. Rao *et al.*<sup>4</sup> synthesized Sr<sub>2</sub>CeO<sub>4</sub>: Eu<sup>3+</sup>, La<sup>3+</sup> phosphor by solid state reaction method, which could be used in white light emitting display devices. Shi *et al.*<sup>5</sup> found that by turning of the concentration of Eu<sup>3+</sup>, Sr<sub>2</sub>CeO<sub>4</sub>: Eu<sup>3+</sup> could provide a foundation for potential application in single-color or two-color fluorescence thermometry techniques. But, Pieterse *et al.*<sup>6</sup> pointed that Sr<sub>2</sub>CeO<sub>4</sub> was difficult to be used as commercial phosphor, because the quantum efficiency (QE) of Sr<sub>2</sub>CeO<sub>4</sub> was less than 50 % at room temperature, which was too low to compete with the presently used materials with a quantum efficiency between 80~90 %. Therefore, how to improve the

luminescence of Sr<sub>2</sub>CeO<sub>4</sub> has become much more attractive and necessary.

Many researches showed that impurity ions could produce a change of symmetry and vibration modes around the luminescence center to result in the enhancement of the luminescence intensity<sup>7-9</sup>. Therefore, in this paper, Sr<sub>2</sub>CeO<sub>4</sub> and impurity ions doped Sr<sub>2</sub>CeO<sub>4</sub>:M (M = Mg, Ca, Zn, Al) phosphors were synthesized by hydrothermal method using (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub> as the precipitation agent. Moreover, phase structure and luminescence properties of the phosphors were investigated by X-ray diffraction and fluorescence spectrophotometry respectively.

### EXPERIMENTAL

Sr(NO<sub>3</sub>)<sub>2</sub>, Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, Mg(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, Ca(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O, Zn(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, Al(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O, (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub>. were used as starting materials. They were all analytical reagents (A.R.).

**Detailed procedure was as follows:** The appropriate stoichiometric proportion of Sr(NO<sub>3</sub>)<sub>2</sub>, Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O and metal nitrate were dissolved in deionized water and fully stirred. Then (NH<sub>4</sub>)<sub>2</sub>CO<sub>3</sub> was dropped slowly into the above solution to adjust the pH value to 9. After 0.5 h of stirring, the mixture was transferred into a teflon-lined stainless steel autoclave with a capacity of 75 %. Hydrothermal reaction was carried out at 180 °C for 6 h and then cooled to room temperature. The obtained suspension was filtered and filter cake washed with distilled water and a little amount of alcohol. The obtained

precursor was put into an oven and dried at 90 °C for 4 h. Finally, the precursor was calcined in a muffle furnace at 1100 °C for 2 h in air.

Phases and crystallization structure of the samples were analyzed and characterized by a Y-2000 X-ray diffractometer using  $\text{CuK}\alpha$  radiation (30 kV  $\times$  20 mA, scanning speed 0.06 °/s and  $\lambda = 1.54178 \text{ \AA}$ ). The excitation and emission spectra of the samples were recorded as pellets on an F-380 fluorescence spectrometer. All samples were performed at room temperature.

## RESULTS AND DISCUSSION

**X-ray diffraction analysis:** Fig. 1 shows the X-ray diffraction patterns of un-doped  $\text{Sr}_2\text{CeO}_4$  and a series of  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn, Al) powders obtained by calcining the hydrothermal precursor at 1100 °C for 2 h. From Fig. 1, it can be seen that nearly all of the peaks of un-doped  $\text{Sr}_2\text{CeO}_4$  and  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn) can be indexed to the  $\text{Sr}_2\text{CeO}_4$ , which agrees well with the JCPDS card (No. 50-0015). According to that, series of samples are orthorhombic phase  $\text{Sr}_2\text{CeO}_4$  with space group Pbam. In these patterns, the peaks of compounds of Mg, Ca or Zn cannot be found, which indicates that  $\text{Mg}^{2+}$ ,  $\text{Ca}^{2+}$  and  $\text{Zn}^{2+}$  ions have entered into the host lattice and have little effect on the crystal structure of the host  $\text{Sr}_2\text{CeO}_4$ .

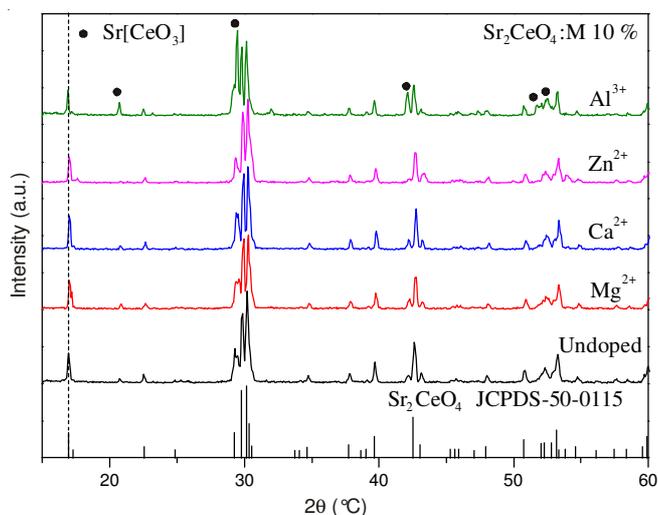


Fig. 1. XRD patterns of un-doped  $\text{Sr}_2\text{CeO}_4$  and  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn, Al) powders

As shown in Fig. 1, the X-ray diffraction peaks of un-doped  $\text{Sr}_2\text{CeO}_4$  are in good agreement with the standard values in JCPDS card (No. 50-0015). However, with the doping of  $\text{Mg}^{2+}$ ,  $\text{Ca}^{2+}$  or  $\text{Zn}^{2+}$ , the position of diffraction peaks of  $\text{Sr}_2\text{CeO}_4$  is deviated to the right, it is because that the smaller radius  $\text{Mg}^{2+}$  ( $R = 0.065 \text{ nm}$ ),  $\text{Ca}^{2+}$  ( $R = 0.099 \text{ nm}$ ) or  $\text{Zn}^{2+}$  ( $R = 0.074 \text{ nm}$ ) ions substitute larger radius  $\text{Sr}^{2+}$  ( $R = 0.113 \text{ nm}$ ) ions in the crystal lattice, which leads to the decrease of lattice parameters and crystal plane spacing. According to the Bragg equation,  $2d\sin\theta = \lambda$ , where  $d$  is the crystal plane spacing,  $\theta$  the diffraction peak angle and  $\lambda$  the radiation wavelength ( $1.54178 \text{ \AA}$  for  $\text{CuK}\alpha$ ), thus the  $\theta$  increases as  $d$  reduces. Therefore, compared with un-doped  $\text{Sr}_2\text{CeO}_4$ , the diffraction peaks of

$\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn) move toward large angle (toward right). However, the doping of  $\text{Al}^{3+}$  ions has little effect on the position of diffraction peaks and the phase of  $\text{Sr}[\text{CeO}_3]$  begins to appear. The possible reason is as follows: the radius of  $\text{Al}^{3+}$  ( $R = 0.05 \text{ nm}$ ) is small enough to occupy any crystal lattice site, it could be located at the sites of  $\text{Sr}^{2+}$  and  $\text{Ce}^{4+}$ , moreover could reside in interstitial sites between or among the host ions. Despite the fact that the radius of  $\text{Al}^{3+}$  is smaller than that of  $\text{Sr}^{2+}$  or  $\text{Ce}^{4+}$ , the size is larger than interstitial site, which results in the cell volume of  $\text{Sr}_2\text{CeO}_4$  becoming larger and affecting the reaction process<sup>10-11</sup>.

**Emission spectra of  $\text{Sr}_2\text{CeO}_4$  and  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn or Al):** Under ultraviolet (UV) radiation at 292 nm, all the samples show bright blue-white emitting. Fig. 2 shows the emission spectra of the samples  $\text{Sr}_2\text{CeO}_4$  and  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  (M = Mg, Ca, Zn or Al) obtained by calcining the hydrothermal precursor at 1100 °C for 2 h. As shown in Fig. 2(c), the emission spectra of  $\text{Sr}_2\text{CeO}_4$  show a broad band in the region 400-600 nm with a peak around 466 nm. The emission band is assigned to the  $f \rightarrow t_{1g}$  transitions of  $\text{Ce}^{4+}$  ions, which is determined by the special one-dimensional chain structure of  $\text{Sr}_2\text{CeO}_4$ . In the  $\text{Sr}_2\text{CeO}_4$  structure, there are two different band lengths of Ce-O in  $\text{CeO}_6$  octahedra, that is Ce-O and Ce-O-Ce and the Ce-O distance is about 0.1 Å shorter than the Ce-O-Ce, which is a critical factor for the observation of charge transfer luminescence. The shape and position of the emission spectra of the metal ion doped  $\text{Sr}_2\text{CeO}_4$  powders are similar to that of the pure  $\text{Sr}_2\text{CeO}_4$  powder, corresponding to the charge-transfer transition from  $\text{Ce}^{4+}$  to  $\text{O}^{2-}$  in the matrix and the result shows that the metal ion doping does not change the luminescence center and the luminescence mechanism of the phosphors. It is found that the doping with  $\text{Al}^{3+}$  or  $\text{Ca}^{2+}$  can reduce the fluorescence intensity of  $\text{Sr}_2\text{CeO}_4$  phosphors, however, the luminous intensity of the  $\text{Zn}^{2+}$  or  $\text{Mg}^{2+}$  doped  $\text{Sr}_2\text{CeO}_4$  powders is found to be enhanced effectively.

For  $\text{Ca}^{2+}$ ,  $\text{Mg}^{2+}$ ,  $\text{Zn}^{2+}$ ,  $\text{Al}^{3+}$  and  $\text{Sr}^{2+}$  ions, the ionic radii are 0.099, 0.065, 0.074, 0.05 and 0.113 nm, respectively. Doping with  $\text{Al}^{3+}$  reduces the fluorescence intensity of the  $\text{Sr}_2\text{CeO}_4$ , there are two reasons about it: one is that  $\text{Al}^{3+}$  ions substitute partially  $\text{Sr}^{2+}$  ions when they enter into the crystal sites, it is a nonequivalent substitution, which will damage the charge balance of the host  $\text{Sr}_2\text{CeO}_4$ ; the other is that  $\text{Al}^{3+}$  ions can easily reside in interstitial sites in the  $\text{Sr}_2\text{CeO}_4$  lattice because of its small size, therefore, the symmetry would be destroyed by the interstitial  $\text{Al}^{3+}$  and the charge-transfer transition from  $\text{Ce}^{4+}$  to  $\text{O}^{2-}$  becomes difficult<sup>11</sup>. Pieterse *et al.*<sup>5</sup> have studied the luminous intensity of Ca-doped  $\text{Sr}_2\text{CeO}_4$  and observed a red shift of the emission, which was attributed to the expansion of the metal-to-ligand bonds in the excited state as a result of the presence of a smaller  $\text{Ca}^{2+}$  ion on a larger  $\text{Sr}^{2+}$  site and the emission spectra relative intensity will decrease. The same phenomenon is also observed in our present study. Doping  $\text{Ca}^{2+}$  ions in  $\text{Sr}_2\text{CeO}_4$  leads to the red-shift of about 4 nm in the emission spectrum and a great decrease of the fluorescence intensity. The size of  $\text{Mg}^{2+}$  and  $\text{Zn}^{2+}$  is smaller than  $\text{Sr}^{2+}$ , but the red-shift in the emission spectra of  $\text{Mg}^{2+}$  or  $\text{Zn}^{2+}$ -doped  $\text{Sr}_2\text{CeO}_4$  powders is not clearly observed and the fluorescence intensity is improved, which can be associated with the larger Pauling electronegativity of  $\text{Mg}^{2+}$  and  $\text{Zn}^{2+}$ .

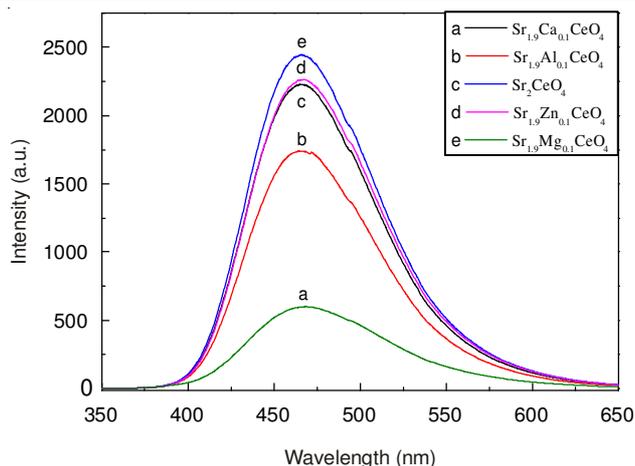


Fig. 2. Emission spectra of un-doped  $\text{Sr}_2\text{CeO}_4$  and  $\text{Sr}_{1.9}\text{M}_{0.1}\text{CeO}_4$  ( $\text{M} = \text{Mg}, \text{Ca}, \text{Zn}, \text{Al}$ ) powders

**Effect of  $\text{Zn}^{2+}$  concentration on luminescence properties of  $\text{Sr}_2\text{CeO}_4$ :** In order to investigate the effect of  $\text{Zn}^{2+}$  concentration ( $x$ ) on luminescent properties of the samples, a series of samples  $\text{Sr}_{2-x}\text{Zn}_x\text{CeO}_4$  ( $x = 0.05, 0.1, 0.15$ ) with different  $\text{Zn}^{2+}$  concentration were synthesized and their excitation spectra are shown in Fig. 3. All the excitation spectra of  $\text{Sr}_{2-x}\text{Zn}_x\text{CeO}_4$  phosphors show a broad band around 286 nm with a shoulder at 336 nm, which is attributed to the charge transfer band of  $\text{Ce}^{4+}\text{-O}^{2-}$ . Compared with pure  $\text{Sr}_2\text{CeO}_4$  powders, a clear red shift occurs in the excitation spectra of Zn-doped  $\text{Sr}_2\text{CeO}_4$  and the main peak is red-shift from 286 nm to 292 nm. We considered that  $\text{Zn}^{2+}$  substitute partially  $\text{Sr}^{2+}$  ions in the crystalline lattice, but  $\text{Zn}^{2+}$  can attract the electrons of  $\text{O}^{2-}$  more strongly than the  $\text{Sr}^{2+}$  because of the larger electronegativity ( $\chi(\text{Zn}^{2+}) = 1.65$  and  $\chi(\text{Sr}^{2+}) = 0.95$ ), which result in decreasing the electron density around  $\text{O}^{2-}$  ion. Thus, the electron transition from the  $\text{Ce}^{4+}$  to  $\text{O}^{2-}$  is liable to carry out when excited and the charge-transfer state band is shifted toward lower energy.

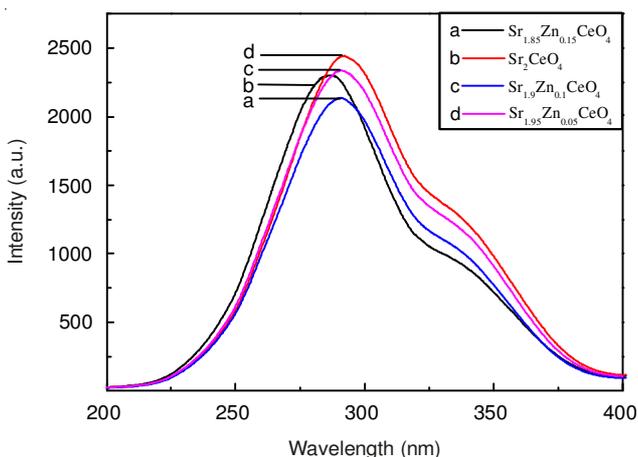


Fig. 3. Excitation spectra of  $\text{Sr}_{2-x}\text{Zn}_x\text{CeO}_4$  powders

Fig. 4 shows that the emission spectra of all the Zn-doped  $\text{Sr}_2\text{CeO}_4$  phosphors excited by the maximum excitation wavelength. It can be seen that the  $\text{Zn}^{2+}$  concentration has little effect on the shape and position of emission peaks, which is located in the range of 400-600 nm with the main peak at 466 nm, but

has great effect on the intensity. It is found that the emission intensity of  $\text{Sr}_{2-x}\text{Zn}_x\text{CeO}_4$  powders increases at first, then decreases with the increase of Zn content. The emission intensity is up to the highest when  $x = 0.05$ . Xue *et al.*<sup>12</sup> have studied the luminescence property of  $\text{Sr}_2\text{CeO}_4$  doped with  $\text{Zn}^{2+}$ . They considered that the possibility of the emission from the  $\text{O}^{2-}$  to  $\text{Ce}^{4+}$  transition is expected to be enhanced in Zn-doped  $\text{Sr}_2\text{CeO}_4$  as a result of the larger electronegativity and the smaller radius of  $\text{Zn}^{2+}$  ions; excessive  $\text{Zn}^{2+}$  addition increase the mutual interatomic interaction, which results in the decreasing of the emission from the  $\text{O}^{2-}$  to  $\text{Ce}^{4+}$  transition.

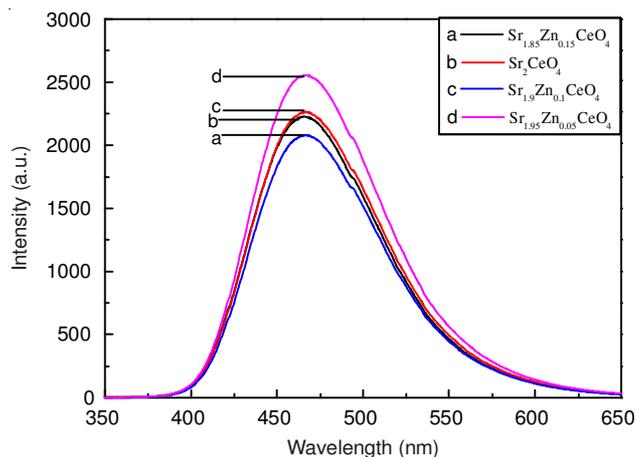


Fig. 4. Emission spectra of  $\text{Sr}_{2-x}\text{Zn}_x\text{CeO}_4$  powders

**Effect of  $\text{Mg}^{2+}$  concentration on luminescence properties of  $\text{Sr}_2\text{CeO}_4$ :** Fig. 5 shows the excitation spectra (monitored at 466 nm) of the samples  $\text{Sr}_{2-x}\text{Mg}_x\text{CeO}_4$  ( $x = 0.02, 0.04, 0.1, 0.15$ ). It can be seen that the concentration of  $\text{Mg}^{2+}$  has little effect on the shape of the excitation spectra, showing a wide band with double peaks in the range of 200-400 nm. Introduction of  $\text{Mg}^{2+}$  ions leads to a red-shift of about 2 nm in the excitation spectra, which is shorter than that of Zn-doped  $\text{Sr}_2\text{CeO}_4$ . The reason is that the electronegativity of  $\text{Mg}^{2+}$  ( $\chi = 1.31$ ) is larger than that of  $\text{Sr}^{2+}$ , but smaller than that of  $\text{Zn}^{2+}$ , which results in the ability of attracting the electrons of  $\text{O}^{2-}$  is higher  $\text{Sr}^{2+}$ , but lower  $\text{Zn}^{2+}$ . Therefore, the red shift of CTS band becomes shorter.

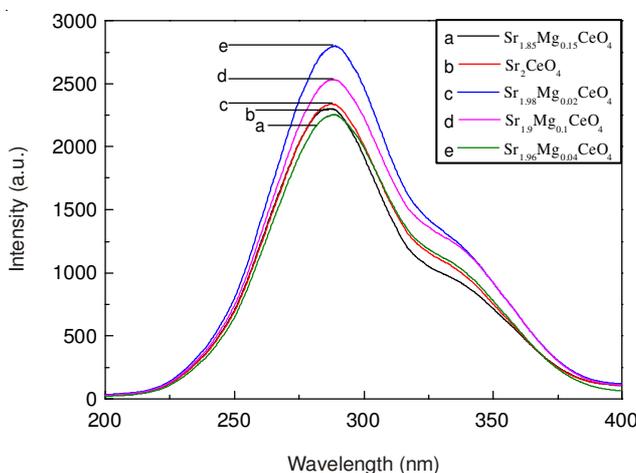


Fig. 5. Excitation spectra of  $\text{Sr}_{2-x}\text{Mg}_x\text{CeO}_4$  powders

Fig. 6 shows that the emission spectra of all the Mg-doped  $\text{Sr}_2\text{CeO}_4$  phosphors excited by the maximum excitation wavelength. It is found that the  $\text{Mg}^{2+}$  concentration has little effect on the shape and position of emission peaks, but has great effect on the intensity. When  $\text{Mg}^{2+}$  doping concentration  $x = 0.04$ , the emission intensity reaches the maximum value. The emission intensity begins to decrease when  $x > 0.04$ . Therefore, it can be known that the appropriate doping amount of  $\text{Mg}^{2+}$  ( $x = 0.04$ ) can improve luminescent intensity of the sample significantly.

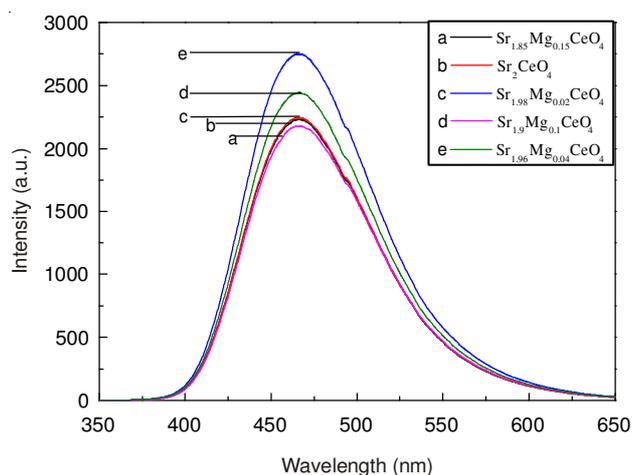


Fig. 6. Emission spectra of  $\text{Sr}_{2-x}\text{Mg}_x\text{CeO}_4$  powders

## Conclusion

Blue-white emitting phosphor undoped and metal ions doped  $\text{Sr}_2\text{CeO}_4$  phosphors were synthesized by hydrothermal method using  $(\text{NH}_4)_2\text{CO}_3$  as the precipitation agent. It is found that  $\text{Mg}^{2+}$ ,  $\text{Ca}^{2+}$  or  $\text{Zn}^{2+}$  doped phosphors can keep the crystal structure of the host  $\text{Sr}_2\text{CeO}_4$  and the position of diffraction peaks of  $\text{Sr}_2\text{CeO}_4$  is deviated to the right. While  $\text{Al}^{3+}$  doped

phosphors have impurity phase of  $\text{Sr}[\text{CeO}_3]$  and the position of diffraction peaks is not deviated to the right. Moreover, the dependences of the photoluminescence on doping species and doping concentration have been investigated. It is found that  $\text{Zn}^{2+}$  and  $\text{Mg}^{2+}$  doping with a appropriate concentration  $x = 0.05$  and  $0.04$ , respectively can effectively enhance the photoluminescence of  $\text{Sr}_2\text{CeO}_4$  phosphors. This investigation provides a new way to enhance the luminescent properties of the  $\text{Sr}_2\text{CeO}_4$  phosphor by using metal ions impurity doping. Thus, the results may be very useful for the application of the  $\text{Sr}_2\text{CeO}_4$  phosphor in future display devices.

## ACKNOWLEDGEMENTS

This study was supported by National Natural Science Foundation of China (No. 50672020).

## REFERENCES

1. G.F. Wang, Q.Y. Mu, T. Chen and Y.D. Wang, *J. Alloys Comp.*, **493**, 202 (2010).
2. J.L. Huang, L.Y. Zhou, Z.L. Wang, Y.W. Lan, Z.F. Tong, F.Z. Gong, J.H. Sun and L.P. Li, *J. Alloys Comp.*, **487**, L5 (2009).
3. E. Danielson, M. Devenney, D.M. Giaquinta, J.H. Golden, R.C. Haushalter, E.W. McFarland, D.M. Poojary, C.M. Reaves, W.H. Weinberg and X.D. Wu, *Science*, **279**, 837 (1998).
4. C.A. Rao, P.R.V. Nannapaneni and K.V.R. Murthy, *Adv. Mater. Lett.*, **4**, 207 (2013).
5. L.L. Shi, H. J. Zhang, C. Y. Li and Q. Su, *RSC Adv.*, **1**, 298 (2011).
6. L. van Pieterse, S. Soerverna and A. Meijerink, *J. Electrochem. Soc.*, **147**, 4688 (2000).
7. M.K. Chong, K. Pita and C.H. Kam, *Appl. Phys. A, Mater. Sci. Process.*, **79**, 433 (2004).
8. K. Park, M.N. Heo, Y. Kim and J.Y. Kim, *J. Nano Res.*, **18**, 257 (2012).
9. W. Yue, J.L. Yan, J.Y. Wu and L.Y. Zhang, *J. Semicond.*, **33**, 073003 (2012).
10. L.L. Shi, C.Y. Li and Q. Su, *J. Alloys Comp.*, **509**, 4209 (2011).
11. T.S. Atabaev, H.H. Thi Vu, Z. Piao, Y.-H. Hwang and H.-K. Kim, *J. Alloys Comp.*, **541**, 263 (2012).
12. S.W. Xue, E.G. Wang and J. Zhang, *Chinese Phys. B*, **20**, 078105 (2011).