

Synthesis of Vinylphenyl Oligomeric Silsesquioxane Based on MQ Silicone Resin

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A novel vinylphenyl oligomeric silsesquioxane (VPOSS) based on MQ silicone resins were synthesised by cohydrolysis and polycondensation of phenyltriethoxysilane, tetraethoxysilane, 1,4-divinyltetramethyldisiloxane and hexamethyldisiloxane. The structure, morphology and properties of the developed copolymers were investigated by FTIR, DSC, XRD, ¹H and ²⁹Si NMR, GPC, TGA and SEM. The heat of fusion (ΔH_f) measured for crystal melting was 21.27 J/g for VPOSS V. The results indicated that the cage and/or ladder structure and the decreasing content of M unit (mol %) in copolymers enhanced the thermal properties of the VPOSSs. The scanning electron microscopy images showed curved surface and lamellar fracture of VPOSS, which can be specialties of potential materials for connectivity and supporter of catalyst. Also, the most important applications are that multifunctional VPOSSs can be used as potential moieties for temperature resistance pressure sensitive adhesive and high performance liquid silicone rubber as light emitting diode encapsulation materials.

Key Words: Silsesquioxane, MQ resin, Polysiloxane, Multifunctional.

INTRODUCTION

Oligomeric silsesquioxane (OSS), $[\text{RSiO}_{1.5}]_n$ ($n = 4, 6, 8, 10, 12, \dots$), with a distinctive nanocage structure consisting of an inorganic core of silicon and oxygen atoms and an outer shell of organic functional groups is one of the most promising nanomaterials for aerospace, electronics, macromers, catalysts, surfaces, biomaterials, nanocomposites and medical applications¹⁻⁶. And these applications were all based on the difference of organic R groups, such as aryl, alkyl, hydrogen, vinyl and other related organic/inorganic groups. As its various functions, phenyl-oligomeric silsesquioxane have received numerous interests in the last few decades^{3-5,7}.

Phenyl-oligomeric silsesquioxane was first reported by Barry *et al.*⁸. And it was used to be synthesized from the hydrolysis and polycondensation of PhSiX_3 , where the X groups being Cl, Br or OR units^{3-5,7,9}. Phenyl-oligomeric silsesquioxane can be modified through various organic or inorganic functional reactions, including acylation, nitration, diazotization and halogenations³.

Moreover, MQ resin, a three-dimensional silicone polymer with low molecular weight, consisted of mono-functional siloxane (M) and tetra-functional siloxane (Q)^{10,11}, have also

absorbed great attention because of its excellent exhibition when it serves as the fillers of temperature resistance pressure sensitive adhesive (PSA)¹² and liquid silicone rubber (LSR) applied for light emitting diode (LED) encapsulant¹³. The so called MQ silicone resin was first synthesized in 1950's¹⁴ and was studied systematically by Huang and his co-workers through cohydrolysis and polycondensation approach¹⁰. In addition, MQ resin also used to be an anchoring backbone for mesogens opens up a way to develop a new group of readily available hybrid liquid crystalline materials¹⁵.

However, above discussed oligomeric silsesquioxanes were always directly prepared with mono kinds of functional, the approach to generate multifunctional oligomeric silsesquioxane derivatives depends on the functionalization of precursor oligomeric silsesquioxane through hydrosilylation of alkenes or alkynes with Si-H group contained oligomeric silsesquioxane^{3,16-18}.

In this paper, we tentatively attempt to directly synthesize a novel kind of multifunctional vinylphenyl oligomeric silsesquioxane (VPOSS) by the cohydrolysis and polycondensation approach. Thereafter, we discussed the structure, thermal properties and morphologies of VPOSS series, which can be specialties of potential materials for connectivity and supporter

of catalyst. The multifunctional VPOSS can also be potential moieties for temperature resistance PSA and fillers for high performance LED encapsulation LSR.

EXPERIMENTAL

Phenyltriethoxysilane (PTES, 98 %) was purchased from J & K Scientific Ltd. (China). Hexamethyldisiloxane (MM, 99 %) and 1,4-divinyltetramethyldisiloxane (^{vi}MM^{vi}, 99 %) were purchased from Zhejiang Sanmen Qianhong Industries Co., Ltd. (Zhejiang, China). Tetraethyloxysilane (TEOS, 99 %) was purchased from Shanghai Lingfeng Chemical Regent Co. Ltd. (Shanghai, China). All of the chemicals were used as received.

Synthesis of VPOSS: VPOSS were prepared by cohydrolysis and polycondensation. The given amounts of monomers (Table-1) were added in a flask equipped with a stirrer and reflux condenser. The HCl solution (1 mol/L) was used to adjust the pH of the reaction system to 1-2 and then the reaction temperature was gradually raised to 75 °C for conducting the cohydrolysis procedure in *ca.* 2.5 h. Whereafter, the toluene, as azeotrope former, was added to the system to remove the low-boiling ethanol-water compounds. The pH of reaction system was adjusted to 10-11 with NaOH for carrying out the postcondensation distillation, which was in alkaline media at 80 °C for *ca.* 1.5 h until no distillate was observed in the distilling head and the toluene-ethanol-water azeotrope was removed from the system simultaneously. The product was cooled to room temperature before analysis.

Purification of VPOSS: To purify the product, deionized water was used to extract ethanol and wash the organic layer to pH 6-7. The upper organic layer was then dried to transparent with anhydrous calcium chloride. After the desiccant was removed by filtration, a solution of VPOSS in toluene with a concentration *ca.* 50 % (wt %) was obtained. A viscous liquid/solid powder was obtained after removing the volatile matters from the solution at 10 mmHg with gradually increasing temperature to 150 °C and kept for 0.5 h.

Characterization: ¹H and ²⁹Si nuclear magnetic resonance (NMR) were used to characterize the structure of VPOSS. ¹H NMR spectra were acquired on Bruker Avance III 500 MHz spectrometer using CDCl₃ as a solvent, while ²⁹Si NMR were carried out on Bruker Avance DPX-300MHz spectrometer with solvent of CDCl₃.

Infrared absorption spectra were collected on a Nicolet PROTÉGÉ 460 Fourier transform infrared (FTIR) instrument. Samples were dissolved in dichloromethane to make a concentration of 10 % (wt %) and coated on a KBr lens or ground with KBr crystal to make 1 wt % mixtures and pelletized.

Gel permeation chromatography (GPC) analysis of all the samples was carried out on PL-GPC 50 Plus system equipped

with a 290-LC pump injection, two 300 mm × 7.5 mm PLgel MIXED-B and MIXED-C columns and a PL-RI Detector. The operation was performed at 35 °C, using tetrahydrofuran (THF) as the solvent with a flow rate of 1.0 mL/min. The molecular weight standards are polystyrenes.

The thermal-transitions of the copolymers were determined on TA instruments Q20 differential scanning calorimetry (DSC), which was calibrated from the melting temperature (T_m) of indium (156.63 °C). The temperature was increased at a rate of 10 °C/min from -80 to 240 °C.

Thermo gravimetric analysis (TGA) measurement was carried out on TA instrument SDT Q600 at a flow rate of 50 mL/min in nitrogen or air. The temperature was increased at a rate of 10 °C/min from room temperature to 800 °C.

The X-ray diffraction (XRD) analysis of the product was performed on Rigaku D/max 2500 PC powder X-ray diffractometer using CuK_α (40 KV and λ = 0.154 nm) radiation, over the 2θ range of 2-90° with a step size of 0.02° at a scanning speed of 0.2 °/min.

The morphology of samples was observed in a field emission scanning electron microscopy (SEM) instrument (JEOL, JSM-6360LA) under an operating voltage of 15.0 kV. Sample powders were sputtercoated with 1-4 nm of Au to reduce charging effects.

RESULTS AND DISCUSSION

Characterization of VPOSS: Generally, the synthesis of oligomeric silsesquioxane lasted for few days³. In present work, we attempted to obtain the expected multifunctional oligomeric silsesquioxane rapidly, therefore the classical cohydrolysis and polycondensation method has been adopted for this purpose. The different molar ratio of feed materials for synthesized VPOSS series were listed in Table-1 and the chemical and thermal analysis were presented in Table-2.

FT-IR spectroscopy: FT-IR spectrum of VPOSS IV is shown in Fig. 1. The relatively broad band at 3425 cm⁻¹ is the intermolecular and/or intramolecular hydrogen bonds (SiOH-H)¹⁹. The strong sharp peaks at 1633 and 958 cm⁻¹ are caused by symmetric deformation vibration of Si-CH=CH₂ associated with M(VI) grafted onto VPOSS¹⁹. While the sharp peak at 1255 and 838 cm⁻¹ are confirmed to be the symmetric deformation vibration of Si-CH₃ in VPOSS¹⁹. In addition, the sharp peaks at 2959, 2853 and 2925 cm⁻¹ are attributed to the stretching vibration of C-H group on -Si(CH₃)₂CH=CH₂ and the peaks at 3008, 3051 and 3074 cm⁻¹ are assigned to the stretching vibration of C-H group on Si-C₆H₅¹⁹. Furthermore, peaks at 1822, 1890 and 1959 cm⁻¹ are associated with the overtone and combination bands of C-H on Si-C₆H₅ deformation vibrations. And, the peaks of 697, 738 and 789 cm⁻¹ are formed by ring deformation vibration and C-H deformation vibration of ring

TABLE-1
FEED MOLE RATIO FOR SYNTHESIS OF VPOSS SERIES

Entry	MM (mmol)	PTES (mmol)	TEOS (mmol)	^{vi} MM ^{vi} (mmol)	R/Si	Status
I	34.568	62.5	39.904	8.871	1.71	Liquid
II	26.543	62.5	39.904	8.871	1.58	Liquid
III	18.519	62.5	39.904	8.871	1.44	Solid
IV	9.259	62.5	39.904	8.871	1.24	Solid
V	0	62.5	39.904	8.871	0.96	Solid

TABLE-2
CHEMICAL AND THERMAL ANALYSIS OF SYNTHESIZED VPOSS SERIES

Entry	R/Si	M _n ^a	M _w ^a	M _w /M _n ^a	T _g (°C) ^b	T _m (°C) ^b	T _{d5%} (°C) ^c
I	1.71	1200	1500	1.21	-60	—	180
II	1.58	1300	1600	1.25	-33	—	228
III	1.44	1500	2100	1.40	-7	—	233
IV	1.24	2600	1.48	43	—	342	
V	0.96	2300	3600	1.57	—	30	401

^aDetermined by GPC. ^bDetermined by DSC, 10 °C/min. ^cDetermined by TGA in air, 10 %/min.

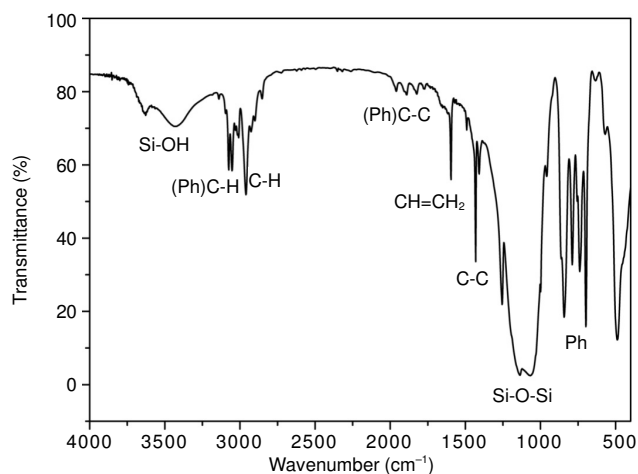


Fig. 1. FT-IR spectra of synthesized VPOSS IV

hydrogens of phenyl group, respectively¹⁹. Moreover, the sharp peaks at 1595 and 1431 cm⁻¹ are C-C stretching vibration of mono-substituted benzene (Si-C₆H₅), which indicates the presence of Si-Ph grafted on VPOSS¹⁹. The classical asymmetric stretching vibration of Si-O-Si is a broad band¹⁹ at 1200-1000 cm⁻¹. All these correspond with the results obtained by ¹H and ²⁹Si NMR analyses.

In addition, we have known from the FT-IR studies²⁰⁻²³, whether the ladder or cage structure, are both possible structure for oligomeric silsesquioxanes. In Fig. 1, the peaks ranging from 1200-1000 cm⁻¹ originate from the stretching vibration of the -Si-O-Si- bond in the horizontal and vertical direction of the polysilsesquioxane backbone, indicating that it is one of the structures of either ladder, cage or random network^{20,22}. Furthermore, two absorptions of bimodal shape appear near 1150 and 1050 cm⁻¹ indicate ladder structure of oligomeric silsesquioxane, while one strong absorption peak near 1100 cm⁻¹ shows closed cage structure^{20,21,24} in FT-IR spectra. Also, the strong bimodal shape can be seen clearly in Figs. 1 and 2, implying that VPOSSs processed a larger portion of the ladder structure than other type of structures shown in Scheme-I.

¹H and ²⁹Si NMR spectroscopy: The structure of VPOSSs was characterized by ¹H and ²⁹Si NMR. A complicated phenyl region at 7-8 ppm with a group peaks is observed in the ¹H NMR spectra (Fig. 3). This implies that the copolymers have the random structures¹⁹. Also, a broad peak detected at 5.3-6.2 ppm are assigned to vinyl group of -OSi(CH₃)₂CH=CH₂. In addition, for sample III, broad weak peaks observed at 3.74 and 1.77 ppm indicate that some residual unreacted Si-OCH₂CH₃ groups still remain in the copolymers¹⁹, while weak peak at 2.36 ppm and hump peak at 0.5-1.4 ppm are assigned to the residual Si-OH groups (Fig. 3).

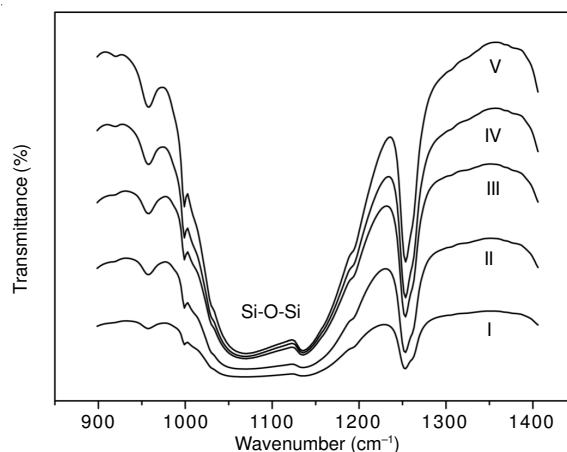
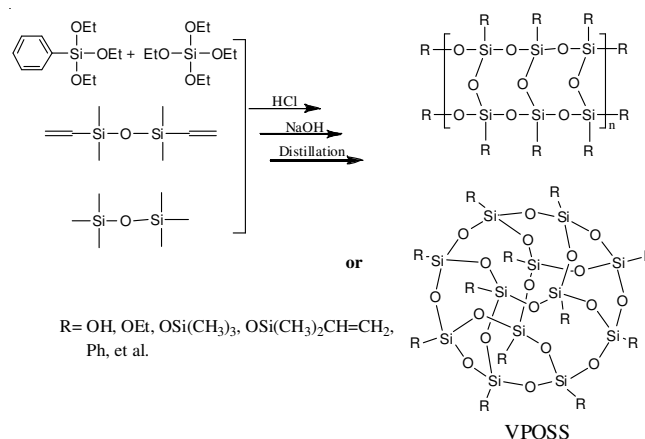


Fig. 2. FT-IR spectra in range of 1400-900 cm⁻¹ for identifying the structure of synthesized VPOSS series



Scheme-I: Synthesis procedure of VPOSS series

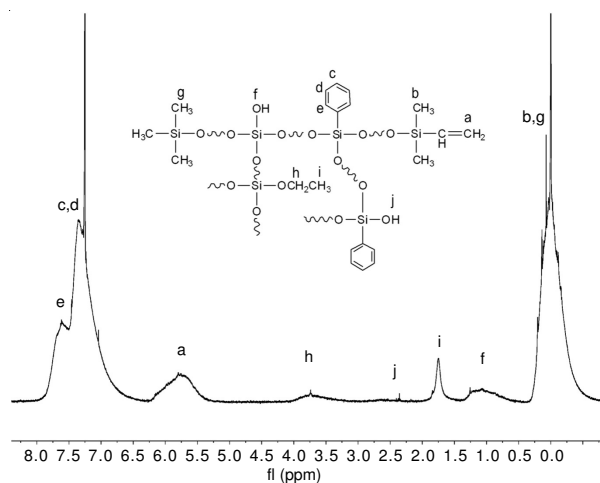


Fig. 3. ¹H NMR of synthesized VPOSS IV

^{29}Si NMR spectroscopy has been considered a useful tool to determine the structure of organosilicon compounds^{4,19,25}. As is shown in Fig. 4, four resonances observed in ^{29}Si NMR spectra can be accounted for M, ^{vi}M , T^{Ph} and Q units, respectively (Figs. 4 and 5)^{19,26}. And the chemical shifts of M, ^{vi}M , T^{Ph} and Q units are 9.62, -2.64, -80.93 and -110.24 ppm, respectively¹⁹. The theoretical content (mol %) of $-\text{OSi}(\text{CH}_3)_3$ is decreasing from I-V (Table-1), fortunately, this trend has been confirmed and seen obviously in Fig. 5, where the area of M unit region of VPOSS at 9.62 ppm reduces gradually. The broad shoulder of these peaks shows that the structure of copolymers are randomized^{4,25,27}.

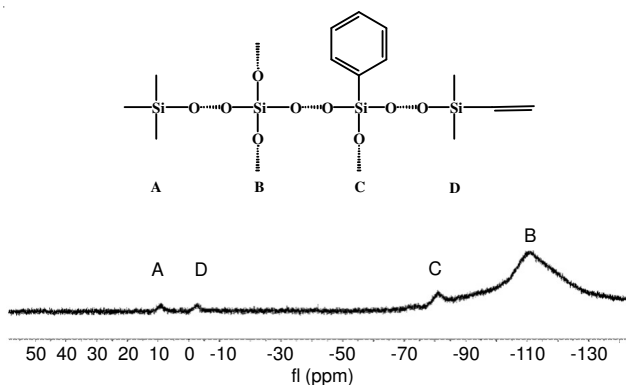


Fig. 4. ^{29}Si NMR of synthesized VPOSS IV

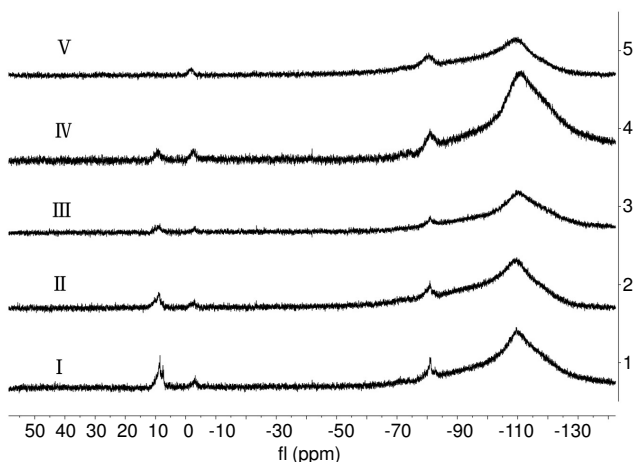


Fig. 5. ^{29}Si NMR comparison of synthesized VPOSS sample series

Thermal properties: The low molecular weight (from 1200-2300 g/mol) of the VPOSSs physically behaved like from high viscosity liquid to solid powder at room temperature. As is shown in Fig. 6 and Table-2, these materials have the T_g (-60 °C to 43 °C) and T_m (ca. 30 °C). The slightly higher glass transition temperature observed for the VPOSS series is attributed to the decreasing content of M units, which can increase the molecular weight and ladder structure of VPOSS. As discussed above, the ladder structure is also observed in FT-IR spectra (Fig. 2). Moreover, the T_m (60-265 °C) for oligomeric silsesquioxane phase was detected by DSC. Melting transitions have been observed from oligomeric silsesquioxanes²⁵. Samples V with a T_m (ca. 30 °C) can be accounted to the VPOSS cages or ladders crystal. In addition, the heats of fusion (ΔH_f), the total heat of fusion over the whole melting transition measured

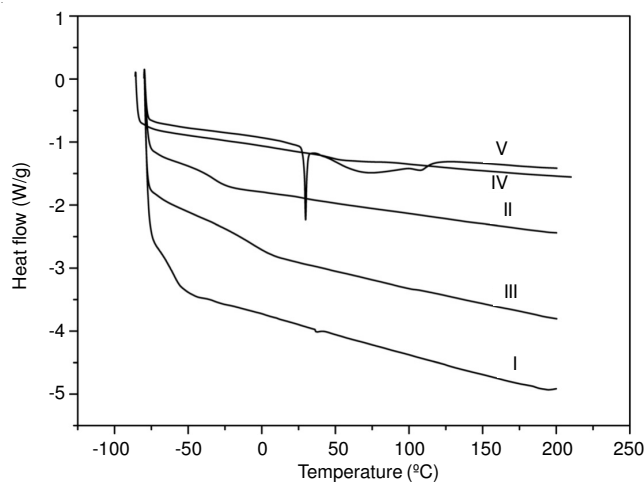


Fig. 6. DSC curves of synthesized VPOSS series, 10 °C/min

for VPOSS V, was calculated from the sharp and broad peaks in DSC curves shown in Fig. 6. And the $\Delta H_{f,V,VPOSS}$ is 21.27 J/g of VPOSS V, which is close to the reported values⁴.

The temperature for the highest thermal stability of all VPOSSs is 401 °C with 5 % decomposition in air (T_d , 5 %, air). Thus, the thermal stability is almost the same or better than that of MQ silicone resin²⁸ (Fig. 7, Fig. 8 and Table-3). As is shown in Fig. 8, the first step of thermal decomposition at range 200-460 °C is probably a consequence of decomposition of organic groups, may be resulted from the polycondensation and degradation of $-\text{Si}-\text{CH}=\text{CH}_2$, $-\text{Si}-\text{Ph}$, $-\text{Si}-\text{CH}_3$, $-\text{Si}-\text{OH}$ or $-\text{Si}-\text{OEt}$, while the second step can be accounted for the degradation of the backbone of VPOSS at range 460-700 °C in nitrogen²⁹⁻³¹. Hence, significantly increased thermal stability compared with that of MQ silicone resin would be expected if the ladder structure oligomeric silsesquioxane were the core and cross-linker of copolymers²⁰. Moreover, the onset temperature increased with the decreasing content of M unit mol % in copolymers indicates that the ladder and/or cage structure can surely enhance the thermal stability of VPOSSs^{20,25}.

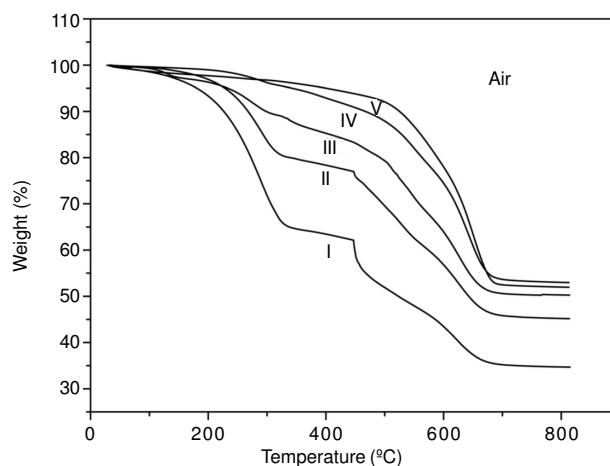


Fig. 7. TGA curves of synthesized VPOSS series in air, 10 °C/min

X-Ray diffraction: XRD analysis used to investigate the VPOSSs structures will present more detail information. Fig. 9 shows the X-ray diffraction of VPOSS IV and V.

TABLE-3
TEMPERATURE AT WEIGHT LOSS OF THERMO GRAVIMETRIC ANALYSIS IN AIR AND N₂

Entry	Temperature at weight loss (%) in air ^a				Temperature at weight loss (%) in N ₂ ^b			
	5	10	15	20	5	10	15	20
I	181	226	253	273	181	224	249	267
II	228	268	293	339	226	268	292	311
III	233	299	407	492	258	302	332	361
IV	319	397	527	564	342	467	529	564
V	401	525	561	590	353	525	572	634

^aObtained from TGA data in air, 10 °C/min. ^bObtained from TGA data in N₂, 10 °C/min.

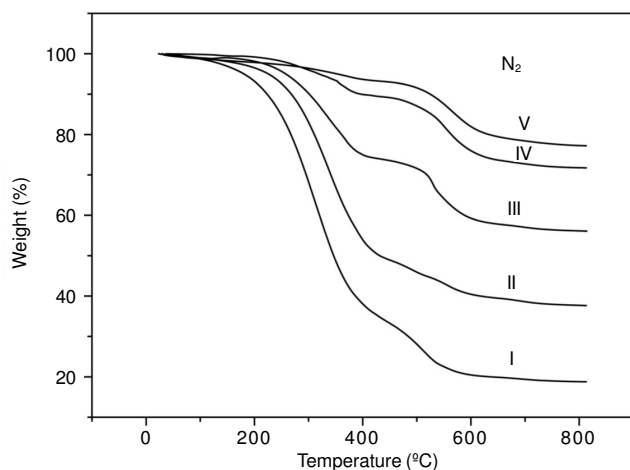


Fig. 8. TGA curves of synthesized VPOSS series in N₂, 10 °C/min

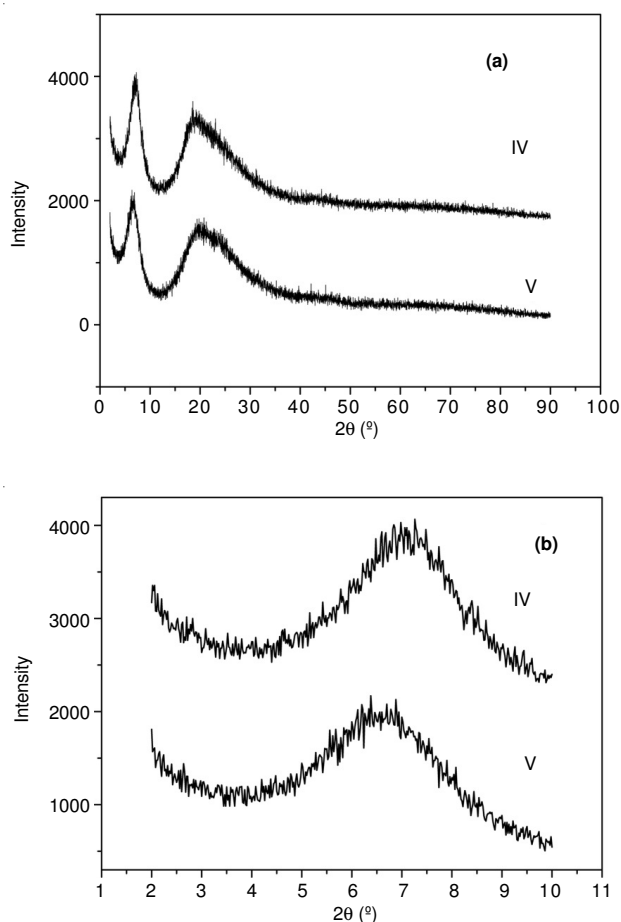
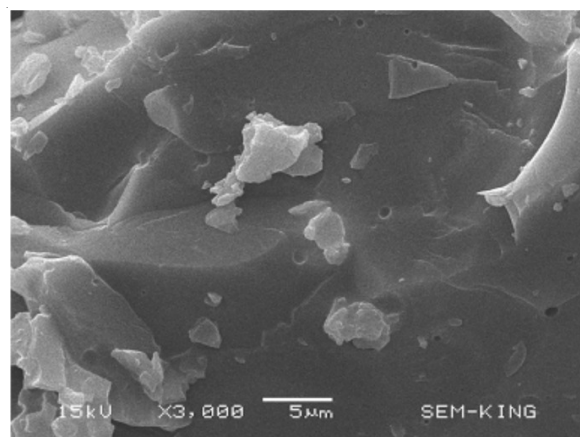
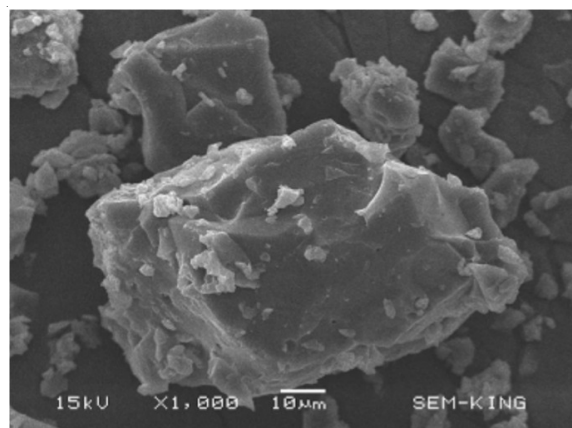


Fig. 9. XRD spectra of synthesized VPOSS IV and V (a) full scale, (b) extended scale

As discussed above, VPOSSs with good thermal properties can tentatively interpret its cage/ladder structure^{3,24,25}. The reaction chemistry is presumably identical for each VPOSS, powder XRD shows peaks 2θ at 6.98° , 18.55° corresponding to a d-spacing of *ca.* 12.65 and 4.77 Å for VPOSS IV, while for VPOSS V is 2θ at 6.66° , 19.88° corresponding to a d-spacing of *ca.* 13.26 and 4.46 Å, respectively. This result is also in excellent agreement with the predicted ladder width (12.2 Å) and ladder thickness (5.1 Å), on the basis of molecular simulation calculations with the simulation software, Alchemy 2000, version 1.0³².

Scanning electron microscopy: The SEM images in Fig. 10 reveal a gross morphology of VPOSS V. The curved surface and lamellar fracture show the disordered, glassy structure of the VPOSS V materials, which is similar to cage structure POSS morphology reported by Roll^{3,24}. It can be the potential materials for connectivity and supporter of catalyst²⁴ and the most important property is the potential usage of fillers in LSR as LED encapsulation materials.



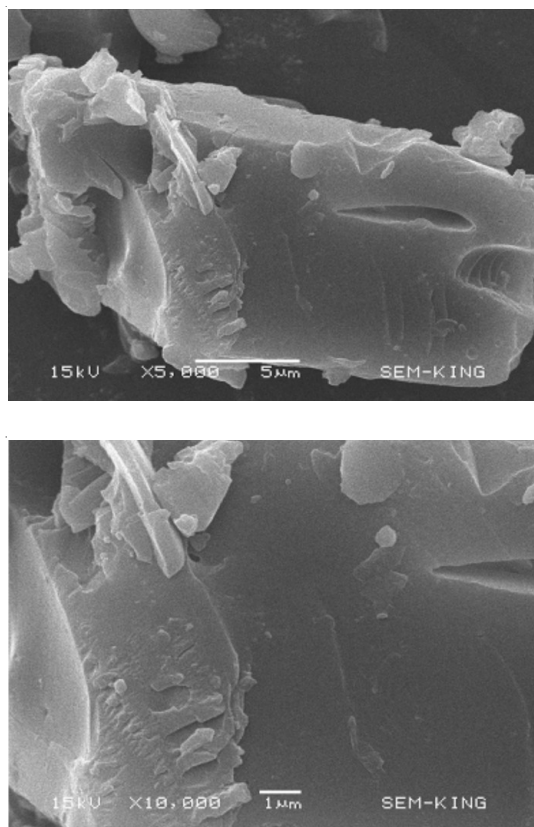


Fig. 10. SEM images of VPOSS V showing evidence of curved surface and lamellar fracture

Conclusion

Vinylphenyl oligomeric silsesquioxane based on MQ silicone resin have been directly synthesized by two steps of cohydrolysis and polycondensation. The cage and/or ladder structure of the copolymers were proved by FT-IR, ^1H and ^{29}Si NMR, XRD. Their molecular weights were range of 1200–2300 g/mol. Thermal transitions at T_g ($-60\sim-43\text{ }^\circ\text{C}$) and T_m (*ca.* $30\text{ }^\circ\text{C}$) were characterized and the heat of fusion (ΔH_f) measured for crystal melting was 21.27 J/g for VPOSS V. In addition, results indicated that the cage and/or ladder structure and the decreasing content of M unit mol % in copolymers enhances the thermal properties of the VPOSSs. And the temperature with the highest thermal stability of all VPOSSs is $401\text{ }^\circ\text{C}$ of 5 % decomposition in air. The curved surface and lamellar fracture observed in SEM images shows a disordered, glassy structure of the VPOSS V, which can be specialties of potential materials for connectivity and supporter of catalyst. And the most important application is that the multifunctional VPOSS can be potential moieties for temperature resistance PSA and high performance LED encapsulation LSR.

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